



particle orientation in response to the external field influence [17]. NF requires a filler with an excellent magnetic response to develop a magnetic field sensor, such as  $\text{Fe}_3\text{O}_4$  nanoparticles. These nanoparticles have superparamagnetic characteristics with good magnetization saturation and interact well with external magnetic fields [18]. The previous study regarding applying  $\text{Fe}_3\text{O}_4$  as a magnetic field sensor reported that the laser intensity increases linearly with increasing the external magnetic field [19]. Improving the performance of  $\text{Fe}_3\text{O}_4$  as a magnetic sensor can be achieved in several ways, such as  $\text{Mn}^{2+}$  doping. Theoretically, this dopant leads to greater magnetization saturation, where the  $\text{Mn}^{2+}$  ion has a greater magnetic moment than  $\text{Fe}^{2+}$  ( $4 \mu\text{B}$ ), namely  $5 \mu\text{B}$ . According to the calculation, the magnetic moment of  $\text{Mn}_x\text{Fe}_{3-x}\text{O}_4$  nanoparticles such as  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  increases with the addition of the  $\text{Mn}^{2+}$  dopant into the  $\text{Fe}_3\text{O}_4$  system [20]. Moreover, another study confirmed that adding  $\text{Mn}^{2+}$  dopant increased the saturation magnetization of  $\text{Fe}_3\text{O}_4$  nanoparticles [19].

Furthermore, the response ability of nanoparticles to a magnetic field depends on the type, shape, and size of nanoparticles. Magnetic materials such as  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  nanoparticles easily aggregate due to van der Waals force, magnetic interactions between particles, and a large surface energy [22]. Aggregation affects and decreases the response to the magnetic field due to the reduced magnetization value. Moreover, aggregation also affects the stability of NF [23], where it causes the structure of the NF to become unstable [24]. Meanwhile, the performance of NF as a magnetic field sensor is good if the stability is high [25]. The aggregation of the NF filler can be minimized by adding a polymer as a template during the synthesis of nanoparticles. Among various existing polymers, polyvinyl alcohol (PVA) is widely chosen as a surfactant because it can produce fillers in dispersed conditions. After all, agglomeration can be prevented, and the PVA coating can prevent the oxidation of magnetite particles [22]. In addition, PVA is composed of hydroxyl groups and a carbon backbone, making PVA hydrophilic. This allows PVA to dissolve in water, resulting in high stability and good chemistry. Furthermore,  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  with a PVA surfactant was coated with a tetramethylammonium hydroxide (TMAOH) as a second surfactant and dispersed in  $\text{H}_2\text{O}$ . TMAOH is a stabilizer that prevents the magnetite nanoparticles in the solution from agglomerating. The combination of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  with PVA and TMAOH surfactants is expected to have high stability and good magnetic field sensor performance. Based on the literature review, PVA plays an important role in preventing aggregation in  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  nanoparticles. However, the optimal composition of PVA volume to obtain  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  nanofluids with high stability is not reported. Therefore, the role of PVA volume variation is important to investigate in order to produce nanofluids with excellent performance as magnetic field sensors. Furthermore, the research was also carried out to investigate the effect of PVA volume on the structure, morphology, magnetic, and optical properties of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF with double surfactant.

## 2. Experimental Section

### 2.1. Synthesis of $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ nanofluid with PVA surfactant

In this study, iron sand was used as the primary material for synthesizing  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ .  $\text{MnCl}_2 \cdot 4\text{H}_2\text{O}$ , hydrochloric acid (HCl), ammonium hydroxide ( $\text{NH}_4\text{OH}$ ), PVA, and TMAH were obtained from Merck, and distilled water pro analysis was used. The  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  nanoparticles with PVA surfactant were synthesized using the coprecipitation method. The first step involved preparing a PVA solution by dissolving 2 g of PVA powder in 100 mL of deionized water. Then, the preparation of  $\text{FeCl}_2$  and  $\text{FeCl}_3$  solutions was conducted from iron sand [27]. Furthermore, 18 mL of the obtained  $\text{FeCl}_2$  and  $\text{FeCl}_3$  solutions were added to  $\text{MnCl}_2 \cdot 4\text{H}_2\text{O}$  and stirred using a magnetic stirrer for 15 min. Then, the solution was titrated with PVA volume variations of 0, 2, 4, 6, 8, and 10 mL, stirred for 30 min, followed by titration of  $\text{NH}_4\text{OH}$  as much as 25 mL for 30 min. After titration, a precipitate of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  nanoparticles with PVA surfactants was formed. Then, the precipitate was washed with  $\text{H}_2\text{O}$  until the pH was neutral, and 1 g was taken to manufacture NF following the research we had reported previously [12]. The  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  precipitate with the PVA surfactant was dissolved in 2 mL of TMAOH and stirred for 1 h at room temperature. Next, the mixture was dispersed in  $\text{H}_2\text{O}$  and stirred for 1 h at 650 rpm at room temperature. Then, the  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  precipitate with PVA surfactant was dried at  $100^\circ\text{C}$  to obtain powder  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  with PVA surfactant.  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  with PVA were then assigned codes P1, P2, P3, P4, P5, and P6, corresponding to  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  with PVA at 0, 2, 4, 6, 8, and 10 mL.

### 2.2. Characterization

Phase analysis was performed using X-ray diffraction (XRD) type X'Pert Pro to determine the quality of the synthesized  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF. XRD characterization was performed to determine the nanoparticle's phase, structure, and crystallite size. Furthermore, the functional groups that make up the nanoparticles were found using Fourier-transform infrared spectroscopy (FTIR), the Shimadzu IR-Prestige 21. In addition, magnetic and optical properties were characterized using a vibrating sample magnetometer (VSM) with the Physical Properties Measurement System Quantum Design PPMS® brand of VersaLab™ Cryogen-free 3 Tesla and UV-Vis Specord 200 Plus, respectively. The performance of NF as a magnetic field sensor was analyzed using an external magnetic field, a laser, and a lux meter.  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF was placed into a container, then externally magnetized on both sides. Furthermore, the container containing the NF was irradiated by a laser to determine the intensity of the light produced by the sample, where the circuit magnetic sensor testing followed our previous research [28].

### 3. Results and Discussion

#### 3.1. Structure Analysis of $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ with PVA surfactants

The XRD patterns of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  nanoparticles with a PVA surfactant as an NF filler are shown in Fig 1. Based on the patterns, it was found that all samples had a peak with the same diffraction, which was detected at  $2\theta$  values of approximately  $30.10^\circ$ ,  $35.50^\circ$ ,  $43.11^\circ$ ,  $56.90^\circ$ , and  $62.41^\circ$ . The diffraction pattern of P1, a pure  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  sample, resembles the peak diffraction of  $\text{Fe}_3\text{O}_4$  in our previous studies, with the highest peak at  $35.65^\circ$  [29]. In comparison, the peaks shifted toward the smaller  $2\theta$ . The shift occurred because Mn, with a radius of  $0.82 \text{ \AA}$ , managed to enter the system and replace some Fe ions with a lower radius,  $0.77 (\text{Fe}^{2+})$  and  $0.65 (\text{Fe}^{3+})$ . Besides that, the addition of Mn does not create new peaks, such as the usual MnO at  $2\theta$  values of  $37^\circ$  (400) and  $64.5^\circ$  (002) [26]. The absence of a new phase was also due to Mn successfully entering the  $\text{Fe}_3\text{O}_4$  system. Next, peak diffraction after the addition of PVA (P2–P6) showed no difference from the peak of P1. These results confirm that PVA successfully acts as a surfactant. However, the addition of PVA affected the intensity of the peak of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ , where the increase in intensity shows the growth of crystals. This is because PVA contains hydroxyl groups capable of forming hydrogen bonds with anions, which causes the solubility of the metal salt to increase crystal growth [31]. Moreover, the PVA polymer can inhibit cation mobility and prevent  $\text{MnFe}_2\text{O}_4$  decomposition, so forming a new phase can be avoided [28].

Furthermore, the color lines in Fig. 1 show the diffraction pattern from the fitting results using the AMCSD database model (No. 0007394). The results of the analysis show that peak diffraction represents planes of (2 2 0), (3 1 1), (4 0 0), (3 3 3), and (4 4 0), with the highest peaks being for (3 1 1) at  $35.50^\circ$ , in accordance with the literature [33]. Furthermore, based on Table 1, it is known that the lattice parameter values ( $a = b = c$ ) of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  with PVA surfactants are in the range of  $8.383 - 8.385 \text{ \AA}$ . These values were much larger than the lattice parameter value of  $\text{Fe}_3\text{O}_4$  from previous studies, which range from  $8.358 \text{ \AA}$  to  $8.361 \text{ \AA}$  [27,29]. The increase in the lattice parameter value indicates the presence of  $\text{Mn}^{2+}$  in the tetrahedral section, indicating that  $\text{Mn}^{2+}$  successfully replaced some of the  $\text{Fe}^{3+}$  atoms in the tetrahedral section. It is known that  $\text{Mn}^{2+}$  has an ionic radius of  $0.89 \text{ \AA}$ , while the ionic radius of  $\text{Fe}^{3+} = 0.64 \text{ \AA}$  and  $\text{Fe}^{2+} = 0.77 \text{ \AA}$ .

The presence of Mn doping on the tetrahedral site of  $\text{Fe}_3\text{O}_4$  is in accordance with the Mössbauer results, which are characterized by an increase in the lattice parameter value with the addition of Mn doping. Therefore, the ion distribution of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  is shown by Equation 1.

$$(Mn_x^{2+}Fe_{x-1}^{3+})_{tetra}(Fe_{x-1}^{2+}Fe_{x+1}^{3+})_{octa} \quad (1)$$

In addition, the fitting results confirm that the nanoparticles had a single phase with a cubic spinel structure and a space group of  $Fd-3m$  same as  $Fe_3O_4$ , as illustrated in Fig. 2. Figure 2 (a) is the structure of  $Fe_3O_4$  which consists of Fe (red sphere) surrounded by six oxygen atoms (blue sphere) or called octahedral site and Fe (orange sphere) which binds four oxygen atoms or often called tetrahedral site. Meanwhile, after Mn doping, it can be seen that in the tetrahedral site, some of the Fe, indicated by the orange sphere, is replaced by a gray sphere, which represents Mn.

Based on Table 1, the lattice parameters of  $Mn_{0.3}Fe_{2.7}O_4$  with PVA surfactants were constant, about 8.38 Å. This is consistent with the results of a previous study on the lattice parameters of manganese ferrite particles with the addition of PVA [34]. Furthermore, the crystal sizes obtained ranged from  $12.5 \pm 1.9$  nm to  $10.5 \pm 1.9$  nm. This is due to the effect of adding PVA, where the distortion of bonds between the distance between atoms and compounds is wider, reducing the crystal size [35]. Changes in crystallite size in the sample indicate the successful effect of adding PVA on particle size and retaining the octahedral morphology of samples, where it showed the ability to control particle design [32].

### 3.2. Functional Groups Analysis of $Mn_{0.3}Fe_{2.7}O_4$ with PVA surfactants

The functional groups of the  $Mn_{0.3}Fe_{2.7}O_4$  with PVA surfactant are represented in Fig. 3. Octahedral Fe–O was detected at a wavenumber of  $657\text{ cm}^{-1}$ , one of the characteristics of the  $Mn_{0.3}Fe_{2.7}O_4$  sample with a PVA surfactant [37,38]. Then, the Mn–O bonds overlapped with Fe–O at a wavenumber of  $400\text{--}450\text{ cm}^{-1}$ , which shows the displacement of the Mn bonds with Fe in the tetrahedral section [35, 36]. Then, the presence of PVA was confirmed due to the presence of the C=C bond at  $1350\text{ cm}^{-1}$ ,  $CH_2$  at  $856\text{ cm}^{-1}$  [41], C–H at  $1502\text{ cm}^{-1}$ , C–C at  $1434\text{ cm}^{-1}$ , and O–H at  $1639\text{ cm}^{-1}$  [38,12]. Interestingly, with increasing the PVA volume as a surfactant, for  $Mn_{0.3}Fe_{2.7}O_4$ , OH absorption at  $1639\text{ cm}^{-1}$  was widened, a characteristic hydroxyl group of PVA. This indicates that the existing hydroxyl group of PVA increased in the samples along with the increase in the PVA volume. Apart from the functional groups of  $Mn_{0.3}Fe_{2.7}O_4$  and PVA, the  $CO_2$  bonds originating from the atmosphere were found at  $2380\text{ cm}^{-1}$  [40], and the O–H bond was at  $3429\text{ cm}^{-1}$ , which is characteristic of O–H bonds originating from water [34,44,45].

### 3.3. Morphology of $Mn_{0.3}Fe_{2.7}O_4$ with PVA Surfactants

The morphology of  $Mn_{0.3}Fe_{2.7}O_4$  with PVA surfactant was characterized using SEM, and the results are shown in Fig. 4. Overall, the nanoparticles were inclined-shaped spherical with a nearly uniform particle size distribution and uniform agglomeration. Agglomeration occurred because of the ratio between the surface area and the large

particle volume, where magnetic dipole interactions and van der Waals forces could stimulate particle agglomeration. Thus, they blocked the formation of a single domain, resulting in inhomogeneity [33]. However, with increasing the PVA volume, the agglomeration of inclined nanoparticles decreased. In addition, the composition of the PVA caused a slowdown in the process. Particle agglomeration and particle homogeneity could be maintained, causing nanoparticles to form more spherical shapes. Based on the quantitative analysis, the particle size distribution ranged from  $33.31 \pm 0.63$  nm to  $23.95 \pm 0.14$  nm, which is close to the results reported previously [41]. The particle size distribution trend of the nanocomposites was consistent with the XRD analysis results. However, it had a value that tended to be greater due to SEM characterization was performed on the sample surface.

### 3.4. Magnetic Properties of $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ with PVA Surfactants

The magnetic properties of the nanoparticles were characterized using VSM and analyzed using the Langevin method concerning susceptibility. The results of the characterization and analysis are shown in Fig. 5. All samples displayed S-shaped curves, indicating that all  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  with PVA surfactants were superparamagnetic. This is also confirmed by the coercivity and magnetization field remanence values in Table 2, where the value is close to zero, and it can be ignored. The main part of the nanoparticles was superparamagnetic. Furthermore, the magnetization saturation of the nanoparticles ranged from  $36.539 \pm 0.004$  to  $46.447 \pm 0.056$  emu/g. The magnetization saturation value increased and decreased according to the crystal size trend from the XRD analysis. The highest saturation magnetization value was observed in sample P3, which was  $46.447 \pm 0.056$  emu/g, while the lowest saturation magnetization value was obtained in sample P6, which was  $36.539 \pm 0.041$  emu/g. This shows that magnetization saturation in the sample was influenced by the crystal size of the  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  with the PVA surfactant. Based on Table 1, it is known that P3 has the largest crystallite size of  $13.4 \pm 2.6$  nm, while P6 has the smallest crystallite size of  $10.5 \pm 1.9$  nm. In accordance with these results, it can be said that the magnetization value decreases as the crystallite size of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  with the PVA surfactant decreases. This is confirmed by previous research, where superparamagnetic nanoparticles with a single domain have saturation magnetization values that are directly proportional to size [46]. The decrease in saturation magnetization value in nanoparticles with smaller sizes is due to the fact that small nanoparticles have a large surface area ratio, so that the spin moments on the surface tend to have random orientations. This random orientation is caused by the exchange interaction that keeps the spins parallel, decreasing due to the number of neighbouring atoms surrounding them being fewer than the spin moments in the core. Random surface spins form subdomains

containing parallel spins but with different orientations in other subdomains, reducing the total spin alignment and resulting in low magnetization values.

$$\frac{M_{s6} - M_{s1}}{M_{s6}} \times 100\% \quad (2)$$

Based on Table 1, it is known that P6 has a saturation magnetization value of  $36.539 \pm 0.041$  emu/g, which is significantly lower than sample P1, a  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  sample without PVA, with a value of  $43.761 \pm 0.048$  emu/g. According to Equation 2, it can be seen that PVA can affect magnetic intensity by 16.5%. Furthermore, from the saturation magnetization value, an approximation can also be made to determine the effect of PVA on the stability of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ . The colloidal stability of magnetic nanoparticles is influenced by dipolar interactions between particles, which are modeled using dipolar interaction energy (Equations 3 – 5).

$$E_{dip} \approx \frac{\mu_0 \mu^2}{4\pi d^3} \quad (3)$$

If,

$$\mu \approx M_s \times V \quad (4)$$

$$Stability \approx \left( \frac{M_{st}}{M_{si}} \right)^2 \quad (5)$$

Then the increase in stability can be calculated by Equation 5, where  $M_{st}$  is the saturation magnetization of the last sample and  $M_{si}$  is the saturation magnetization of the first sample. In the calculation, the saturation magnetization value of samples P1 and P6 is used. The calculation results show that the stability after the addition of PVA increases to 1.43 times, indicating that the nanofluids are more stable after the addition of PVA.

### 3.5. Optical Properties of $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ with PVA Surfactants

Fig. 6 shows the direct band gap calculated using Tauc's method. The gap energy of P1, sample  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ , was 3.085 eV. The energy gap was similar to that in the previous study, where the resulting energy gap was around 3.01 eV [47]. A previous study reported that the band gap of  $\text{Fe}_3\text{O}_4$  is 2.3 eV, which is significantly lower than that of sample P1. Mn can increase impurities in semiconductor materials to increase the band gap [48]. This is in line with previous studies where the band gap value of the  $\text{Fe}_3\text{O}_4$  increased with the addition of other materials and the doping process, ranging from 0 to 4 eV [49]. Furthermore, after adding the PVA surfactant, P2, the band gap value increased by 3.301 eV. This increase is due to the addition of PVA, allowing the material to become more insulating and enhancing the band gap value. Detailed band gap values for the  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  are presented in Table 2. The band gap values after the addition of PVA appear to fluctuate but are larger than that of P1. This may also be related to the particle size, where the band gap value is inversely proportional to the particle size [50].

Apart from the band gap, Table 2 shows the refractive index value of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  with a PVA surfactant. The refractive index is related to the ability of light to propagate or penetrate a substance. The refractive index value plays an important role in the development of optical-based technology, especially optical-based magnetic sensors. The refractive index value is related to the intensity of the laser that can penetrate the NF. The highest refractive index value is shown in sample P1, which is 2.135. This result is in line with previous research, which reported a refractive index of 2.2 [51]. Meanwhile, after the addition of PVA, the refractive index value of the sample tends to be lower than that of P1. This can be attributed to the lower density of the nanoparticles after the addition of PVA (Table 1). Furthermore, the trend of the refractive index value was inversely proportional to the band gap value, where the smaller the band gap value, the larger the refractive index value.

### 3.6. Sensor Performance of $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ NF with PVA

The sensor performance of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF with PVA surfactant was investigated according to the influence of laser light with a magnetic field applied to the NF. Fig. 7 represents the resulting transmission optics curve, which gives an external magnetic field at a determined distance. This optical transmission occurs because the laser light is passed through the  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF in a glass beaker, where the  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF generally has isotropic properties, i.e., uniform in all directions, and can absorb only a small amount of laser light when the intensity is high [52].

In this study, the  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF samples were diluted using  $\text{H}_2\text{O}$  at a ratio of 1:2. Then, we tested them using a magnetic field in the range of 0–82.5 mT with external magnetic distances of 3, 2.5, 2, 1.5, 1, and 0.5 cm. A greater magnetic field was detected by decreasing the distance. The light intensity enhancement along the magnetic field distance increased near the magnitude of the magnetic field applied to the NF, thereby increasing the ability to absorb laser light. In pure  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF without PVA, the obtained light intensity was 0.5 lux; along with the increase in PVA surfactant, the light intensity increased but then decreased. The highest light intensity was observed in sample P4. This is because the P4 sample has a lower refractive index value than the others, namely 1.876 (Table 2). Therefore, the laser will easily penetrate the NF liquid from P4. It is known that the refractive index value is a reflection of the optical density of a liquid, where the smaller the refractive index value, the lower the optical density of the liquid and the laser will more easily penetrate the liquid. Additionally, the decrease in refractive index can also be attributed to polarizability. Previous research has demonstrated a linear relationship between the refractive index value and the polarizability value [53]. PVA is included in the low-polarizability materials because it has –OH functional groups that are not easily polarized. The more PVA solution is added, the more –OH functional groups cover the  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  particles, which restrains electron mobility on



the surface and decreases the polarizability of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ , ultimately impacting the decrease in the refractive index value. The smaller the density of a substance will allow light entering the fluid to continue straight away. This phenomenon is explained by the Kerr effect, in which the laser light intensity under the influence of an external magnetic field has a nonlinear effect on the refractive index [54]. Previous studies have stated that the relationship between the laser light intensity, which increases linearly as a function of the external magnetic field, has excellent potential as an optical-based magnetic field sensor. Thus, the study results of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF-based magnetic field sensors have great potential for further development with improvements in various aspects.

#### 4. Conclusion

$\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF with the PVA surfactant was successfully synthesized. The  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  filler with the PVA surfactant had a cubic spinel structure with lattice parameters of 8.383–8.385 Å. The addition of PVA surfactants did not affect the phase and structure of the nanoparticles; however, the effect on the intensity peak increased. It was noticeable that the growth of the crystals of the nanoparticles was improved. FTIR analysis showed that the main functional groups were Fe–O octahedral and Mn–O and Fe–O tetrahedral at wavenumbers of 657 and 400–450  $\text{cm}^{-1}$ , confirming the presence of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$ . The presence of PVA was indicated by the presence of C=C at 1350  $\text{cm}^{-1}$ ,  $\text{CH}_2$  at 856  $\text{cm}^{-1}$ , and C–H at 1502  $\text{cm}^{-1}$ , C–C at 1434  $\text{cm}^{-1}$ , as well as the O–H bond at 1639  $\text{cm}^{-1}$ . All samples had superparamagnetic properties with magnetization saturation of 46.447–36.539 emu/g. The band gap value of the  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  filler increased with the addition of the PVA surfactant, reaching 3.085–3.504 eV. Furthermore, as the band gap increased, the refractive index decreased, which is very influential for optical-based sensor applications. The performance of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF with PVA surfactant as a magnetic field sensor was indicated by the magnetic field curve against light intensity, where the light intensity increased at closer distances between the magnetic field and the NF. This shows that the magnitude of the external magnetic field that hit the NF was larger at a closer distance. Furthermore, light intensity after adding PVA was greater than that of NF without PVA, i.e., up to 2.1 lux. Based on these results,  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF with a PVA surfactant can be applied as a magnetic sensor.

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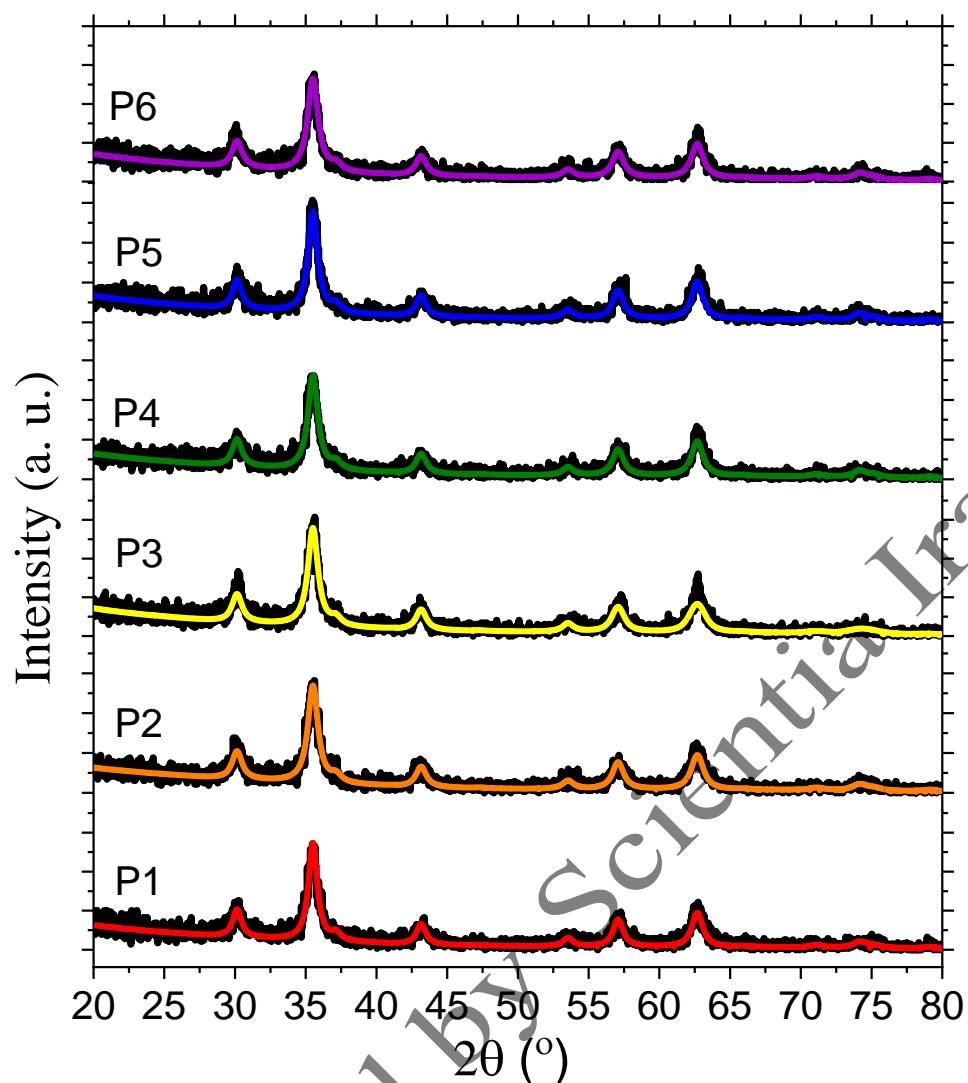
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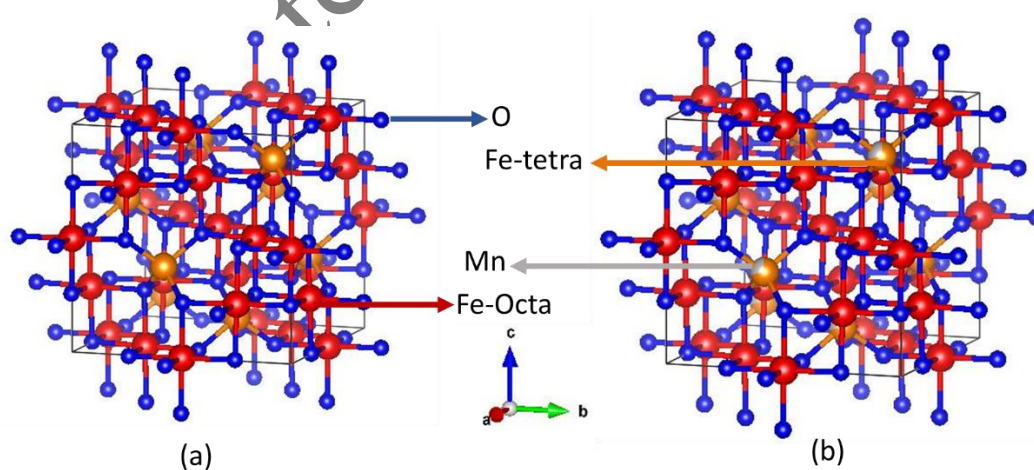
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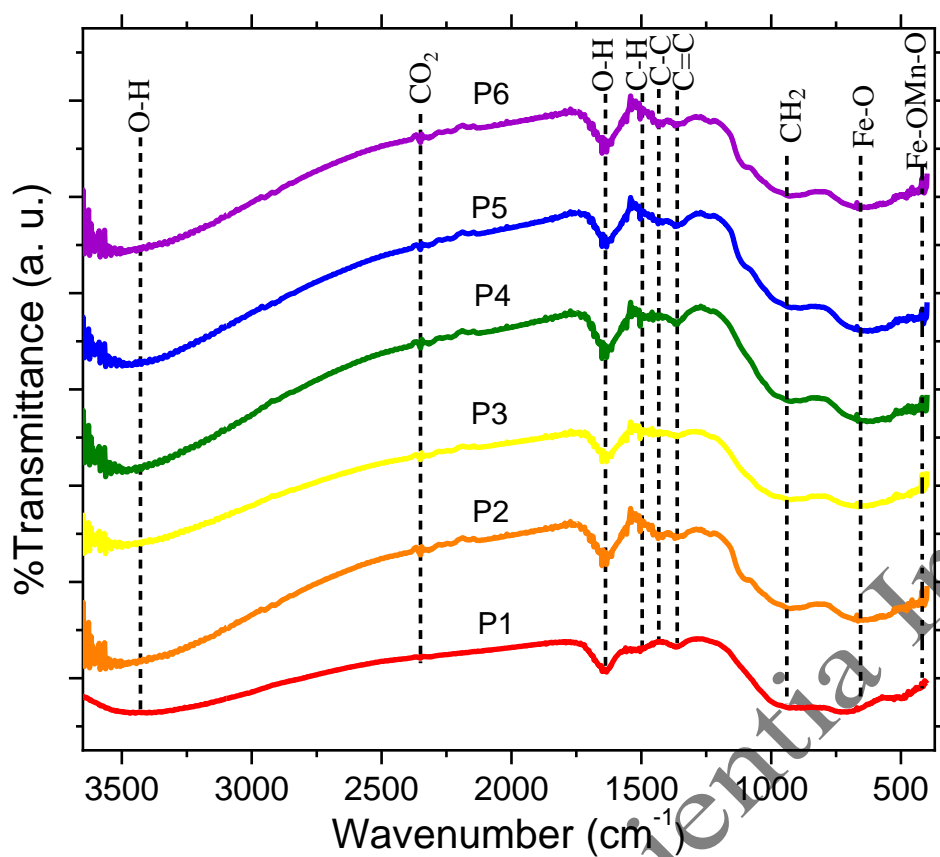
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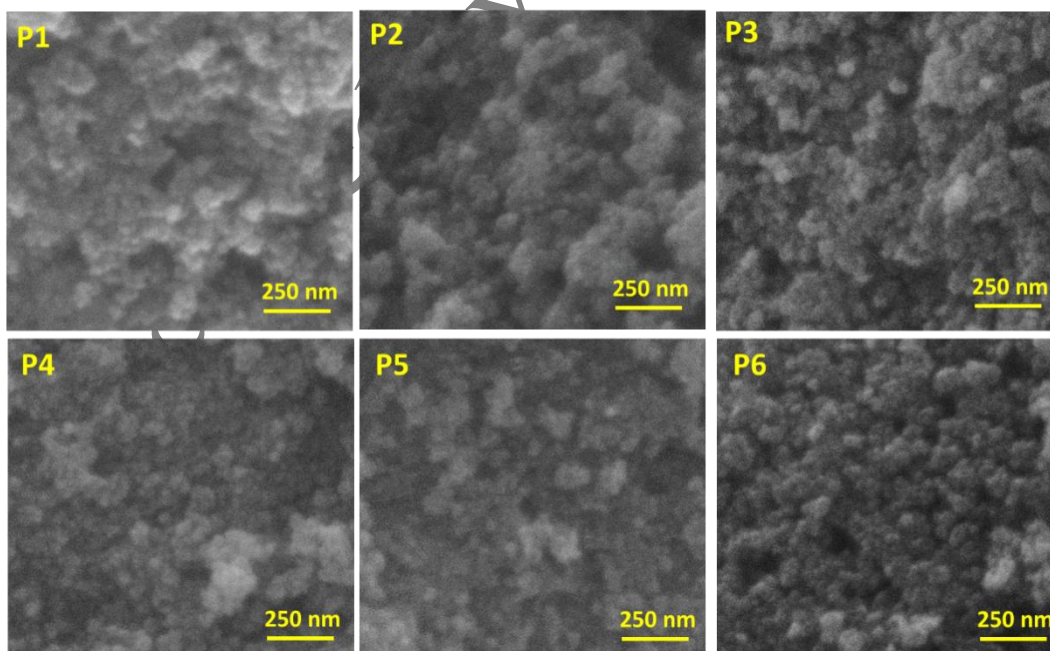
**Fig. 1** XRD diffraction patterns of Mn<sub>0.3</sub>Fe<sub>2.7</sub>O<sub>4</sub> nanoparticles with PVA volume variations



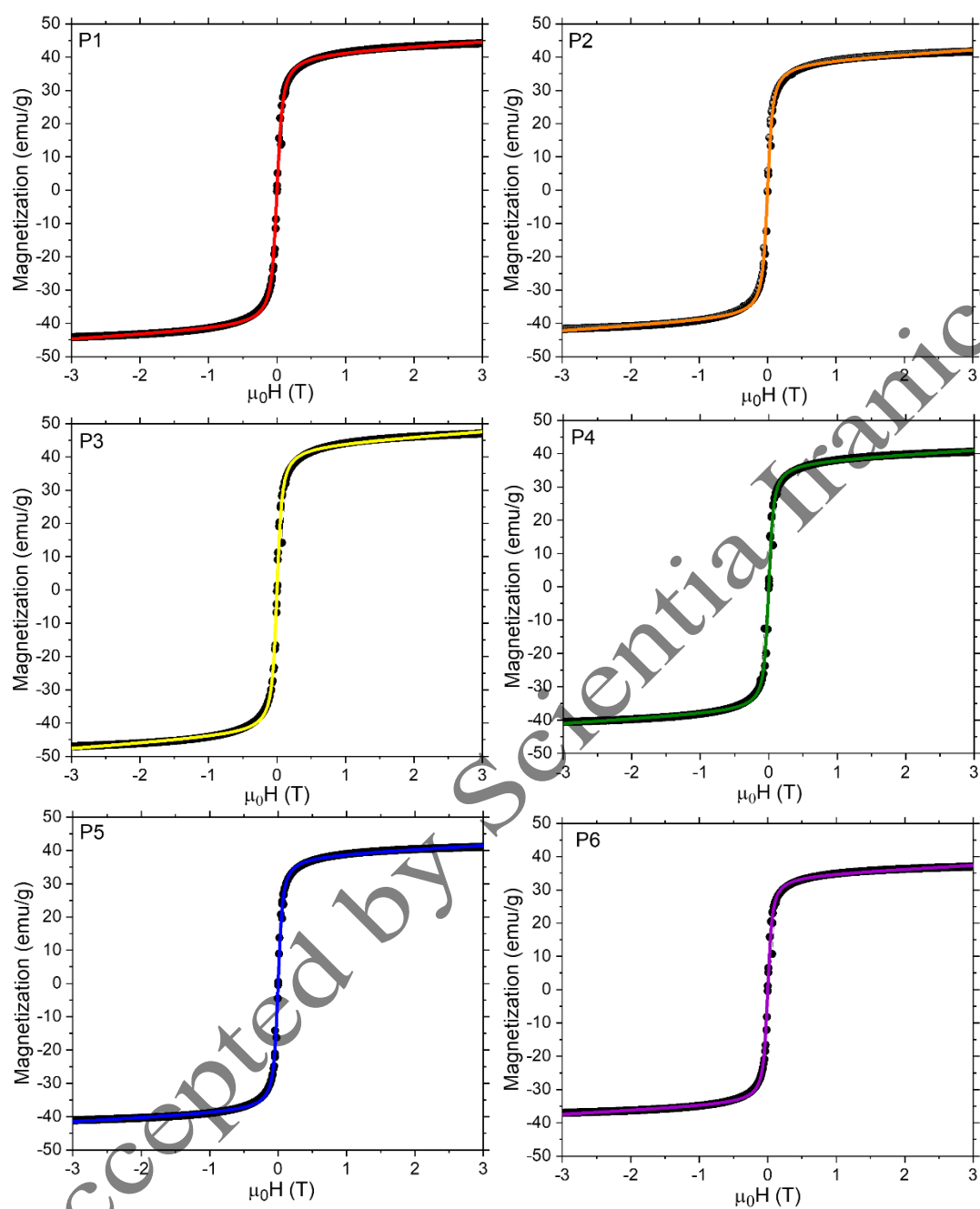
**Fig. 2** Crystal structure of (a) Fe<sub>3</sub>O<sub>4</sub> and (b) Mn<sub>0.3</sub>Fe<sub>2.7</sub>O<sub>4</sub>



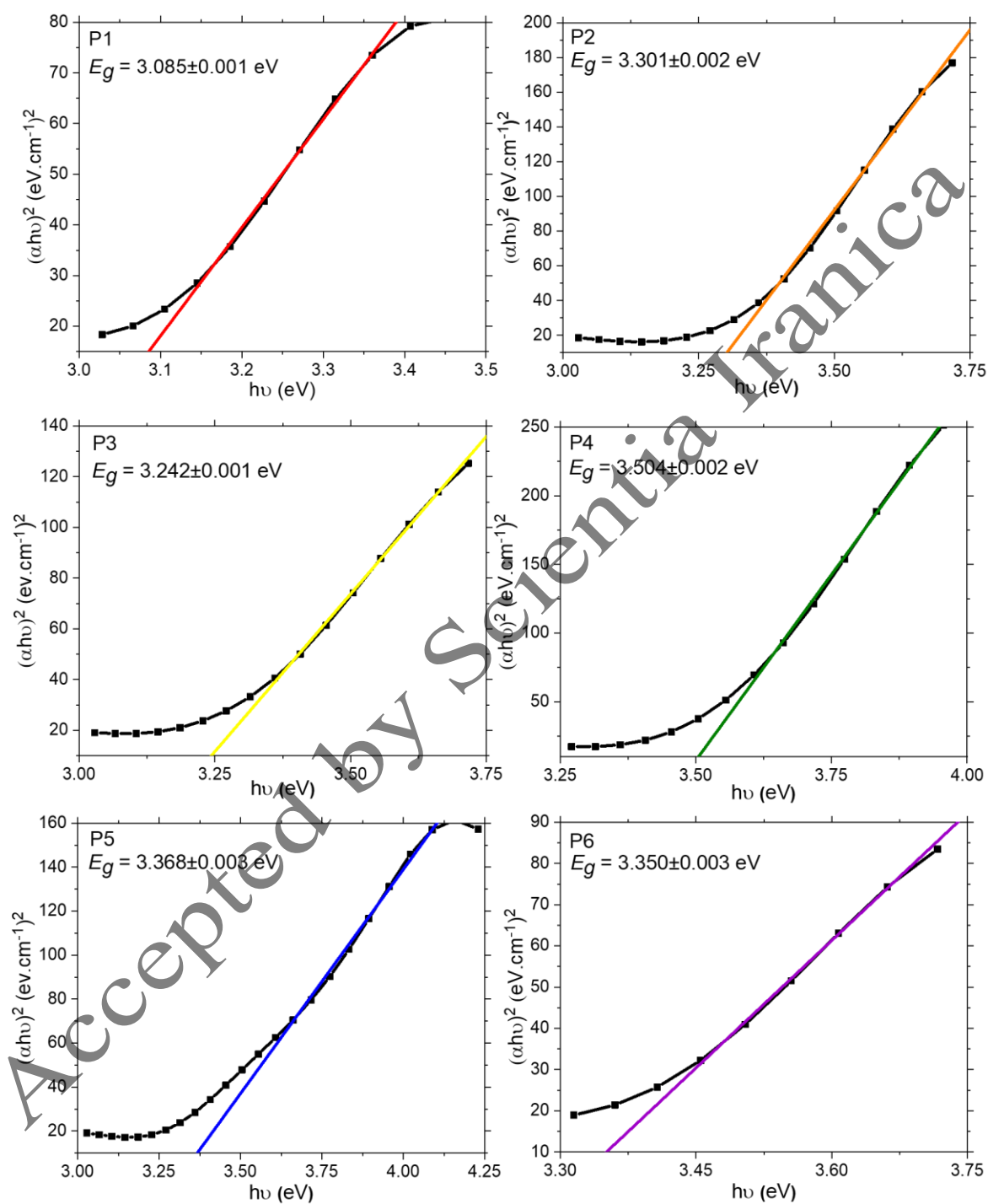
**Fig. 3** FTIR spectra of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  nanoparticles with PVA volume variations



**Fig. 4** SEM images of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  nanoparticles with PVA volume variations

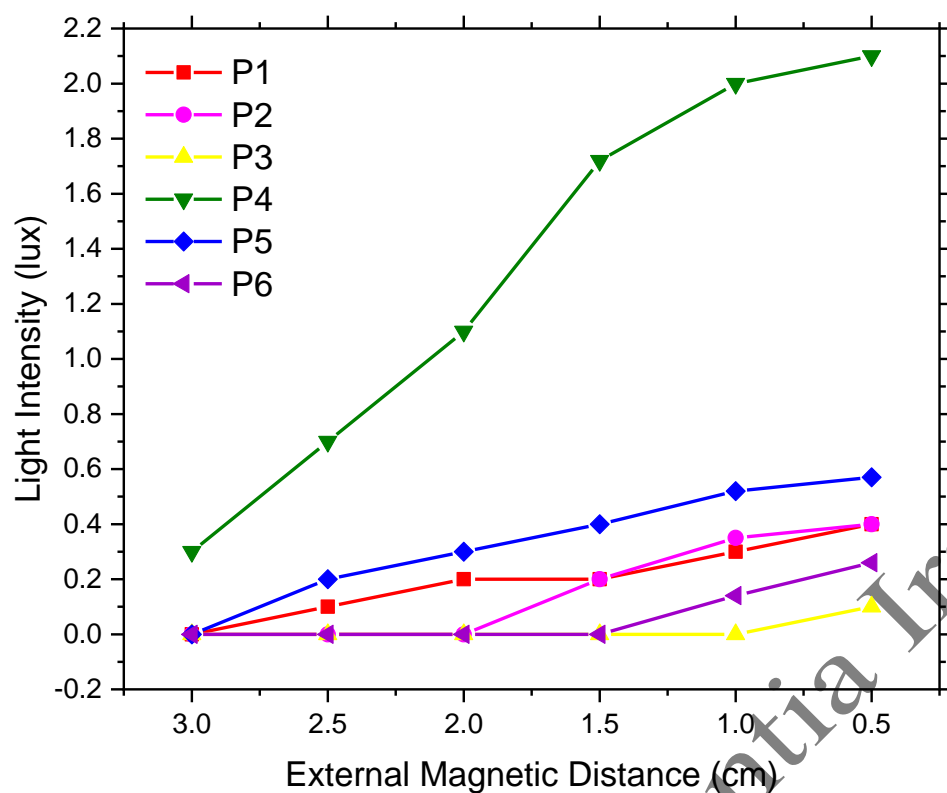


**Fig. 5** Hysteresis curves of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  nanoparticles with PVA volume variations



**Fig. 6** The direct band gap of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  nanoparticles with PVA volume variations





**Fig. 7** Sensor performance of  $\text{Mn}_{0.3}\text{Fe}_{2.7}\text{O}_4$  NF

Samples	$a = b = c$ (Å)	Crystallite size (nm)	Density	Highest peak ( $^{\circ}$ )	Bragg R- factor
P1	$8.383 \pm 0.002$	$12.5 \pm 1.9$	5.210	35.50	9.60
P2	$8.384 \pm 0.001$	$11.4 \pm 1.8$	5.189	35.50	8.90
P3	$8.383 \pm 0.001$	$13.2 \pm 2.6$	5.199	35.50	8.60
P4	$8.384 \pm 0.002$	$10.9 \pm 1.9$	5.128	35.50	7.84
P5	$8.384 \pm 0.001$	$11.1 \pm 2.2$	5.147	35.50	8.16
P6	$8.385 \pm 0.001$	$10.5 \pm 1.9$	5.152	35.52	8.52

**Table 2** Magnetization saturation value ( $M_s$ ), magnetization remanence ( $M_r$ ), field coercivity ( $H_c$ ), and susceptibility ( $\chi$ )

Samples	$M_s$ (emu/g)	$M_r$ (emu/g)	$H_c$ (T)	$\chi$
P1	$43.761 \pm 0.048$	$0.059 \pm 0.030$	$0.001 \pm 0.001$	$1.273 \pm 0.036$
P2	$41.154 \pm 0.046$	$0.002 \pm 0.001$	$0.002 \pm 0.001$	$1.327 \pm 0.029$
P3	$46.447 \pm 0.056$	$0.038 \pm 0.028$	$0.001 \pm 0.000$	$1.485 \pm 0.042$
P4	$40.067 \pm 0.039$	$0.027 \pm 0.002$	$0.002 \pm 0.001$	$1.239 \pm 0.039$
P5	$40.627 \pm 0.032$	$0.017 \pm 0.013$	$0.003 \pm 0.001$	$1.081 \pm 0.026$
P6	$36.539 \pm 0.041$	$0.038 \pm 0.041$	$0.002 \pm 0.001$	$1.077 \pm 0.031$

Sample	Band gap (eV)	Refractive index
P1	3.085	2.135
P2	3.301	2.001
P3	3.242	2.038
P4	3.504	1.876
P5	3.368	1.959
P6	3.350	1.971

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**uAhmad Taufiq** is a Professor and the Head of the Department of Physics at Universitas Negeri Malang. In 2023,

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