



Sharif University of Technology  
**Scientia Iranica**  
*Transactions B: Mechanical Engineering*  
www.scientiairanica.com



# Conditioning of vitrified and resin bond CBN grinding wheels using a picosecond laser

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Received 15 May 2014; received in revised form 25 April 2016; accepted 26 September 2016

## KEYWORDS

Picosecond Yb:YAG laser;  
Vitrified bond;  
Resin bond;  
CBN grinding wheel;  
Laser conditioning.

**Abstract.** Laser ablation is a novel non-mechanical wheel preparation method for optimizing the treatment costs of superabrasive tools. In this study, the thermal effects of picosecond laser radiation on vitrified and resin bond CBN superabrasive grinding wheel surfaces were analytically and experimentally investigated. The analytical approach was intended to find threshold process parameters for selective ablation of cutting grains and bond material. A picosecond Yb:YAG laser device was integrated with a cylindrical grinding machine, which facilitated the treatment of grinding wheel as it was mounted on the grinding spindle. It is shown that the extent of material ablation is defined by the maximum surface temperature induced by the laser radiation, which is in turn defined by the laser pulse energy. It is also suggested that the depth of laser thermal effects is governed by the relative speed of the laser scanner with respect to the wheel surface.

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## 1. Introduction

The development of superabrasive grinding tools has revolutionized the grinding process as a crucial part of manufacturing technology. Nevertheless, the conventional conditioning of superabrasive tools is challenging owing to their high hardness and, accordingly, high dressing tool wear and cost, long process time, and pollution, which have hindered the overall acceptance of the superabrasive tools in industry [1,2]. Non-mechanical wheel preparation methods such as ELID (electrolytic in process dressing), electro-discharge dressing, and laser dressing are novel approaches to

reducing the treatment costs of superabrasive tools. Despite the high initial investment, these techniques are economical in long time due to their positive features like no conditioning tool wear, high speed and controllability, flexibility, environmental friendliness, and limited surface damage.

The first attempts at studying the effects of laser on grinding wheel surface were made in late 1980s, where the laser dressing of conventional alumina wheels was investigated [3,4]. The radiation of the laser beam on the wheel surface can cause the local melting of the bond material and/or the cutting grains. The laser radiation can also help the conventional mechanical dressing methods by the local softening of the wheel material ahead of the dressing tool [5]. In addition to high repeatability, laser dressing provides favorable grinding forces, wheel wear, and ground surface conditions in comparison with the mechanical contact methods [3,4,6]. The benefits are more pronounced when applying the method to

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superabrasive wheels, which are associated with higher costs and more conditioning challenges [7-10]. It is also demonstrated that, owing to higher controllability and smaller heat-affected zone, the application of pulsed lasers is more promising than that of Continuous Wave (CW) lasers [11]. While retaining the same grinding performance as that in grinding with conventionally dressed wheels, laser dressed metal bond diamond wheels are subject to less grain damage and less surface crack propagation [12]. Thermal analysis of the cooling process in laser assisted dressing of alumina wheels suggests the generation of preferable multi-faceted grains, which favors the cutting action [13,14]. Touch dressing of electroplated diamond wheels is also made possible using an ultra-short pulsed picosecond laser [15,16]. The application of ultra-short pulsed lasers for the conditioning of superabrasive tools has also been reported in literature [17,18].

Topographical analysis of the grinding process has shown that only a very small portion of the static cutting edges are involved in the cutting action (dynamic cutting edges) while the rest are engaged in the rubbing and ploughing regimes, doing no material removal [19]. The ratio of the dynamic to the static cutting edges is reported to be in the range of 2% to 12% [20]. The ratio could be increased by reducing the number of momentarily engaging cutting grains while increasing the average chip thickness. A well-known method to perform this task is to diminish the number of static cutting edges via segmentation of the grinding wheel [21]. Regarding the high controllability, repeatability, and precision of the grinding wheel laser treatment, the structuring process can also be conducted. Therefore, a wide range of applications from conventional conditioning (dressing and truing) to structuring can be performed with the same concept over a wide variety of grinding wheel types.

Considering the ability of the laser treatment to replace the conventional grinding wheel preparation methods, and to generate arbitrary structures on the grinding wheel surface, a methodology is proposed to perform the whole range of processes in terms of laser conditioning. Therefore, the truing, dressing, and structuring of the grinding wheel can be flexibly done with the same equipment when comprehensive understanding of the underlying process is achieved. As the laser treatment is a thermal process, thermal behavior of the laser-radiated surface is investigated. Owing to different thermal properties of the grinding wheel components, the selective treatment of cutting grains or the bond material would be possible. Accordingly, the threshold laser power values are found for arbitrary treatment of the grinding wheel components. The effect of laser scanning speed on the conditioning process is also investigated. Owing to its ultra-short pulse duration ( $<10$  ps), a Yb:YAG picosecond laser

device is chosen and integrated with a cylindrical grinding machine. This integration facilitates the accomplishment of in-process laser treatment. The processing time would also be shortened by eliminating the grinding wheel handling and assembly time. Accordingly, laser ablation tests are conducted to assess the efficiency of the method in practice and verify the theoretical background.

## 2. Laser conditioning equipment

A Yb:YAG TruMicro5050 picosecond laser device is used in the current research for thermal conditioning of CBN grinding wheels. The exit beam of the laser with 5 mm diameter cannot be readily radiated to the surface. In addition, to be transferred to the point of interest, the beam should be focused with an optical scanner. The principle of laser structuring is schematically presented in Figure 1, where the beam is guided through some mirrors from the laser head to the laser scanner.

The laser scanner is mounted on a vertical stage, which can position the scanner in the vertical direction. The other required motion components (two horizontal and one rotational) are provided by the grinding machine spindle. The motion components are presented by arrows in Figure 1. Vitrified and resin bond CBN grinding wheels with  $64 \mu\text{m}$  average grain size were chosen for laser treatment experiments. The laser scanner, the focusing lens, and a grinding wheel installed in the grinding machine are presented in Figure 2.

Depending on the optical lens in use, the distance between the laser scanner and the grinding wheel surface is adjusted according to the focal length of the lens. Performing the laser radiation in the focal point provides the highest energy concentration and the finest zone of treatment for selective treatment of the grinding wheel components.

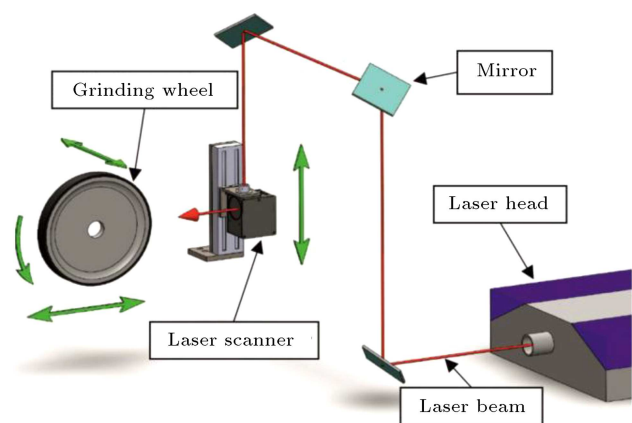
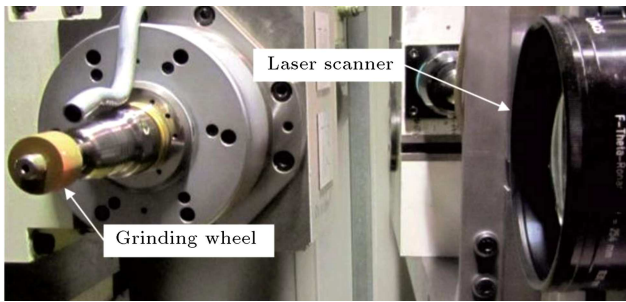


Figure 1. Grinding wheel conditioning concept.



**Figure 2.** Laser scanner and grinding spindle combination.

### 3. Thermal analysis of laser conditioning

Laser conditioning is a thermal process in which a focused high-energy laser beam melts or evaporates every solid material at its focal point. The overall laser treatment can be resolved into the action of individual pulses. Owing to the ultra-short duration of a picosecond laser pulse, the convective heat loss from the grinding wheel surface is negligible. Therefore, only the heat conduction within the grinding wheel body is considered in the present analysis.

The thermal conduction in a solid medium is governed by the heat conduction equation (known as Fourier model), which has the following general form for isotropic materials [22]:

$$\rho c \frac{\partial T}{\partial t} = \nabla \cdot (k \nabla T) + q'' \tag{1}$$

The equation describes the temperature,  $T$ , with respect to time,  $t$ , and spatial coordinates, where  $\rho$ ,  $c$ , and  $k$  are the density, specific heat capacity, and thermal conductivity of the material, respectively. The last term,  $q''$ , is the internal heat generation per unit volume of the material. Considering the laser beam as a point source of thermal energy, which is small in comparison with grinding wheel size, the grinding wheel would be assumed as a semi-infinite half space. The laser pulse would raise the temperature of a point on the surface of the wheel to a maximum value in a short time ( $<10$  ps) after which the heat would be dissipated inside the medium. The thermal properties of the wheel components are presented in Table 1 as they would be addressed later.



**Figure 3.** Laser radiated zone generated by a steady beam at the focal point.

Before going further in simplifying the heat conduction equation, the irradiation of laser pulse on the grinding wheel surface would be considered. The laser device is adjusted at 100% power corresponding to 125  $\mu\text{J}$  pulse energy (to get the highest effect) and 400 kHz repetition rate, and the beam is released at a point on the wheel surface. The picture of the generated zone is shown in Figure 3, where its radius is measured to be about 40  $\mu\text{m}$ .

The laser pulse incidence on the surface can be considered as a thermal disturbance in a short pulse width,  $t_0$ , which takes the local surface temperature up to  $T_{\text{max}}$ . The distance a change of temperature would propagate during a pulse duration can be defined by the thermal diffusion length, which is expressed as [25]:

$$2L = 2\sqrt{\alpha t_0}, \tag{2}$$

where  $\alpha$  is the thermal diffusivity of the material and is equal to  $k/\rho c$ . As the pulse width of the laser is less than 10 ps, the thermal diffusivity length values for a CBN grain, vitrified bond, and resin bond material would be 0.17  $\mu\text{m}$ , 0.02  $\mu\text{m}$ , and 0.0025  $\mu\text{m}$ , respectively, which are much smaller than the average grain size and the beam diameter in both cases. Therefore, the heat transfer during a single pulse can be assumed to take place just in one material (either the grain or the bond material). Considering the grain and bond

**Table 1.** Thermal properties of CBN grains and the vitrified and resin bond material [1,23,24].

Parameter	CBN	Vitrified bond	Resin bond
Density ( $\text{kg/m}^3$ )	3480	3625	1100
Specific heat ( $\text{J/kgK}$ )	506	840	1200
Thermal conductivity ( $\text{W/mK}$ )	1300	30	0.21
Thermal diffusivity ( $\text{m}^2/\text{s}$ )	$7.4 \times 10^{-4}$	$9.8 \times 10^{-6}$	$1.59 \times 10^{-7}$
Melting temperature ( $^\circ\text{C}$ )	3227	1500-2500	302-497

material as homogenous bodies, and eliminating the internal heat generation term, the heat conduction equation can be simplified as:

$$\frac{\partial T}{\partial t} = \alpha \nabla^2 T. \quad (3)$$

The solution to Eq. (3) for a cooling semi-infinite body on the points along the axis exactly under the radiation zone can be expressed as [26]:

$$T(z, t) = T_{\max} + (T_i - T_{\max}) \operatorname{erfc}\left(\frac{z}{2\sqrt{\alpha t}}\right), \quad (4)$$

where  $T_i$  is the initial body temperature (before the radiation of laser beam),  $T_{\max}$  is the surface temperature corresponding to the thermal disturbance, and  $z$  is the position along the axis facing inside the body from the surface.  $\operatorname{erfc}(z/L)$  is the complementary error function whose tabular values are available [26]. Applying the thermal balance to the material contained in a cylindrical surface with a diameter equal to the pulse diameter,  $d_p$ , and a depth equal to  $L$ ,  $T_{\max}$  for a single pulse can be expressed as:

$$T_{\max} - T_i = \frac{2\epsilon W_{\max}}{\pi k d_p^2} \sqrt{\alpha t_0}, \quad (5)$$

where  $\epsilon$  is the absorptivity of the radiated material (about 0.02 for grinding wheel ceramic components [1]) and  $W_{\max}$  is the maximum instantaneous pulse power (12.5 MW). Substituting the proper values and assuming  $T_i$  at room temperature (20°C),  $T_{\max}$  would be about 5223°C for CBN gains and above 8226°C for the vitrified bond. These high temperatures will not happen in practice and the material would melt or vaporize before that. As the experiments will be done in presence of a high-pressure air nozzle, both melting and vaporization will lead to material removal. As the melting temperature of CBN is about 3227°C, the pulse energy would be enough to remove both CBN grains and bond material from the wheel surface. The different maximum temperature values in grinding wheel components suggest that it is possible to selectively remove bond material without melting CBN grains. This selection of parameters would be a favorable choice for dressing applications. It can be concluded from Eq. (5) that there is a somehow linear relationship between the maximum power and maximum temperature attained in the material. According to Eq. (5), the threshold average laser power values for the removal of CBN grains, vitrified bond, and resin bond are about 19 W, and 12 W, and 1 W, respectively.

As the radiation time is too short, only the heat conduction during the cooling process would be considerable. The cooling or relaxation process takes place until the radiation of the next laser pulse at  $t_1$ , which is defined by the pulse repetition rate. The

temperature at the end of cooling process can be expressed as:

$$T(L, t_1) = T_{\max} + (T_i - T_{\max}) \operatorname{erfc}\left(\frac{L}{2\sqrt{\alpha t_1}}\right). \quad (6)$$

Rearranging Eq. (6), we have:

$$\frac{T(L, t_1) - T_{\max}}{T_i - T_{\max}} = \operatorname{erfc}\left(\sqrt{\frac{t_0}{t_1}}\right). \quad (7)$$

As the ratio  $t_0/t_1 = 4 \times 10^{-6}$  for the picosecond laser is very small, the right side of Eq. (7) approaches unity ( $\operatorname{erfc}(0.002) \approx 0.998$ ). In other words, the initial temperature of the next pulse would be the same as the initial body temperature. The stability of the steady laser beam in Figure 3 is also in accordance with this prediction. In fact, the diameter of the heat affected zone remains constant in time.

In addition to the pulse energy and timing, the depth of laser effects depends on the time period during which every specific zone of the grinding wheel surface is subjected to the laser beam. The longer the radiation time, the deeper the surface effects. Therefore, the depth of penetration is proportional to the laser scanning speed. Assuming the depth of laser treatment  $d$  as the accumulation of the material removed per unit pulse, the following relation can be suggested in terms of laser beam scanning speed,  $\nu$ , and laser repetition rate,  $f_r$ :

$$d = \tau(W_{\max}) R L f_r / \nu. \quad (8)$$

where  $\tau(W_{\max})$  is, in general, an empirical function of pulse energy. In combination with the effective laser beam radius,  $R$ , measured in Figure 3, the size and the shape of the radiated area can be approximated as a cylinder of 40  $\mu\text{m}$  radius whose depth is equal to  $d$ .

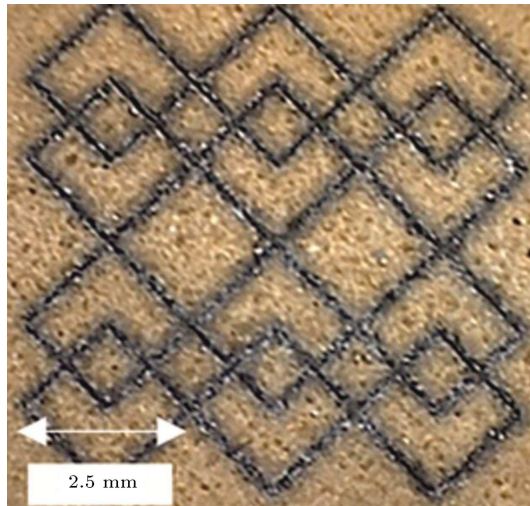
#### 4. Laser ablation experimental results

As the pulse energy defines the maximum generated temperature, various power values would be used to reflect the effect of various surface temperatures, which are also different for grains and bond material. Following the predicted threshold average power values (19 W, 12 W, and 1 W for the CBN grains, vitrified bond, and resin bond materials, respectively) and the fact that the extent of heat diffusion would be controlled by the pulse repetition rate and the scanning speed, a set of experiments were conducted. The pulse repetition rate was kept constant at 400 kHz for all the tests. A set of average laser power values and two sets of scanning speed values were chosen for the experiments. Table 2 presents the chosen sets of parameters.

An air nozzle was used to take the debris and the molten material away from the radiation zone. In

**Table 2.** The parameters assigned for laser conditioning experiments.

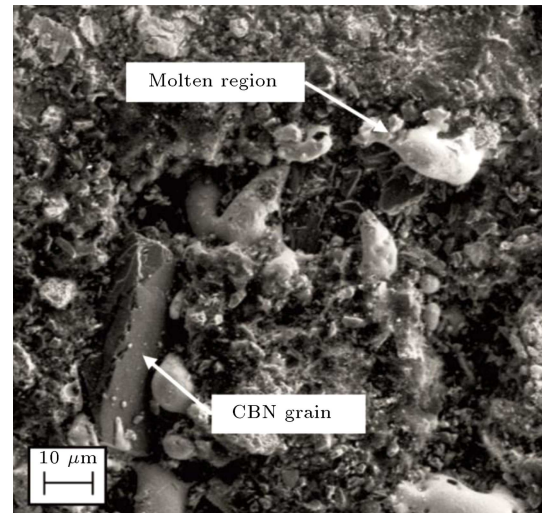
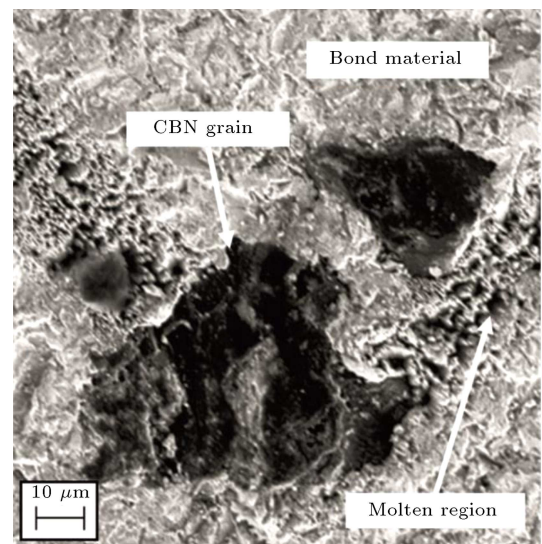
Parameter	Values
Average laser power (W)	2, 10, 15, 20, 25, 50
Scanning speed for vitrified bond CBN wheel (mm/s)	5, 10, 20, 40
Scanning speed for resin bond CBN wheel (mm/s)	100, 150, 200

**Figure 4.** The generated pattern for the laser conditioning tests.

order to have constant cooling and wiping effect, the pressure (7 bar), distance, and orientation of the nozzle were kept constant. The pattern shown in Figure 4 is generated on the grinding wheel surface for further surface investigations and measurements.

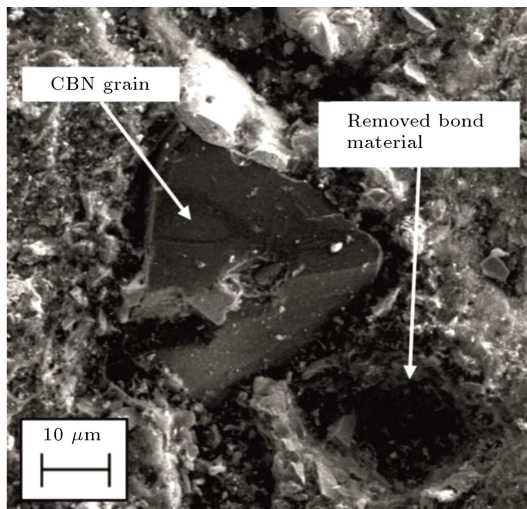
In order to investigate the ablation of the grinding wheel components qualitatively, a more profound study of the radiated surfaces would be needed. Accordingly, SEM images of the vitrified and resin bond samples are taken. The laser treatment of resin bond CBN grinding wheel with 2 W average laser power and 150 mm/s scanning speed is shown in Figures 5 and 6 presents the trace generated with 15 W average laser power ( $37.5 \mu\text{J}$  single pulse energy) and 10 mm/s scanning speed on vitrified bond CBN grinding wheel. As it can be seen, the bond material is slightly molten in both cases, while the cutting grains remain unchanged. Some cavities can be seen, which can be due to the removal of some parts of the bond material with the air stream. Therefore, 2 W and 15 W are the threshold laser power values suggested by the experiments for the removal of resin and vitrified bond, respectively. The experimentally achieved threshold values agree qualitatively with the predictions of the thermal analysis, but the individual values are larger in the experiments. This deviation can be associated with the energy loss of the laser beam through the optical equipment and the blockage of the laser beam just before incidence on the wheel surface by the ablated debris.

By increasing the laser power, the treatment of

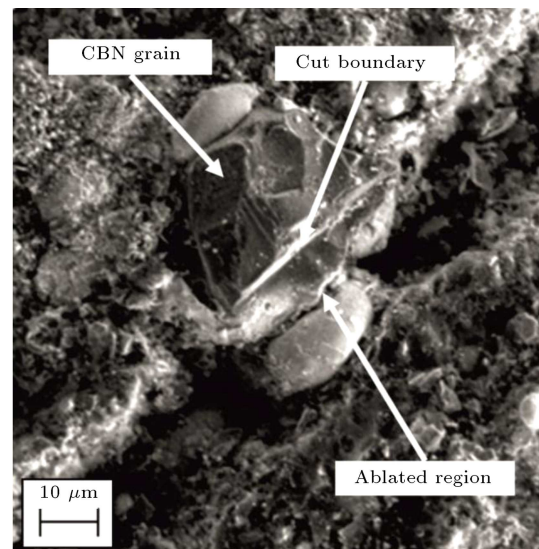
**Figure 5.** SEM image of laser treatment on resin bond CBN wheel with 2 W average laser power ( $5 \mu\text{J}$  single pulse energy) and 150 mm/s scanning speed.**Figure 6.** SEM image of laser treatment on vitrified bond CBN wheel with 15 W average laser power ( $37.5 \mu\text{J}$  single pulse energy) and 10 mm/s scanning speed.

the bond material becomes more significant. Up to 20 W average laser power ( $50 \mu\text{J}$  pulse energy), the bond material in both resin and vitrified CBN grinding wheels can be removed effectively while the CBN grains are not still ablated. Figures 7 and 8 present the traces of laser treatment on resin bond and vitrified bond CBN wheels with 20 W average laser power, and

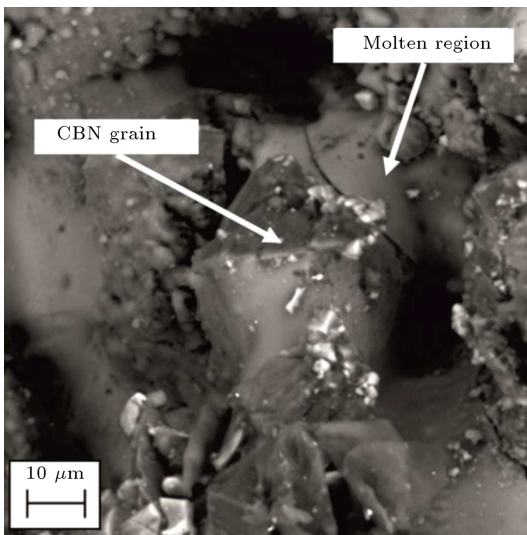




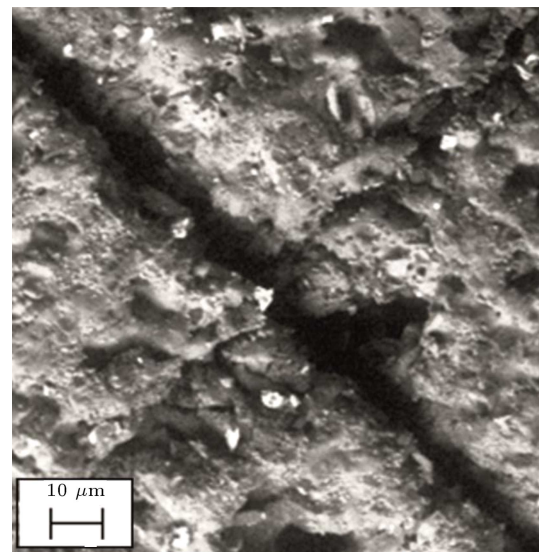
**Figure 7.** SEM image of laser treatment on resin bond CBN wheel with 20 W average laser power ( $50 \mu\text{J}$  single pulse energy) and 150 mm/s scanning speed.



**Figure 9.** SEM image of laser treatment on vitrified bond CBN wheel with 25 W average laser power ( $62.5 \mu\text{J}$  single pulse energy) and 10 mm/s scanning speed.



**Figure 8.** SEM image of laser treatment on vitrified bond CBN wheel with 20 W average laser power ( $50 \mu\text{J}$  single pulse energy) and 10 mm/s scanning speed.



**Figure 10.** SEM image of laser treatment on vitrified bond CBN wheel with 50 W average laser power ( $125 \mu\text{J}$  single pulse energy) and 10 mm/s scanning speed.

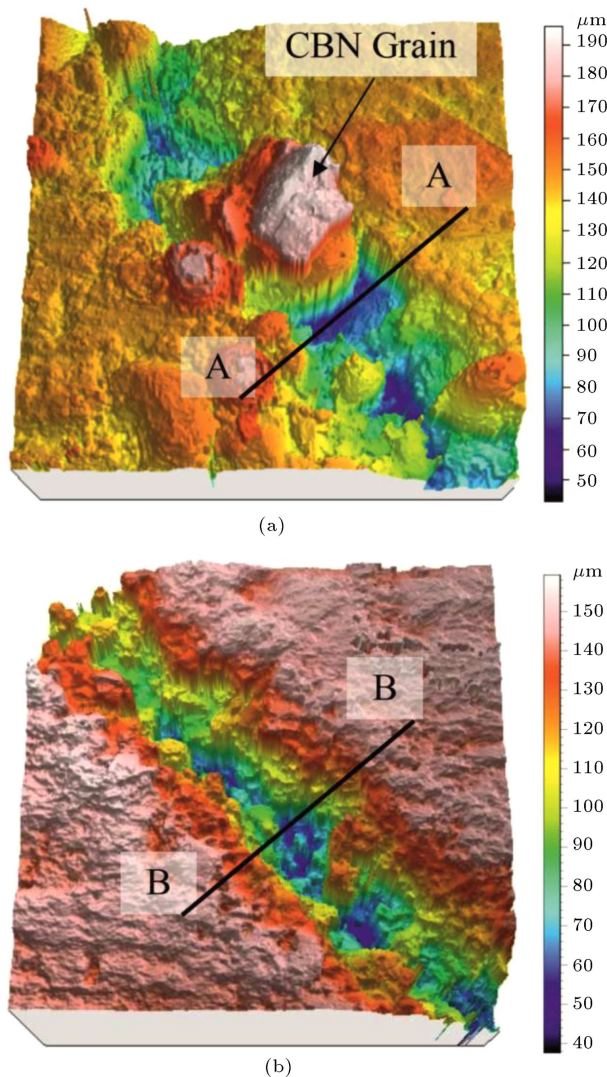
150 mm/s and 10 mm/s scanning speeds, respectively. Some cracks can be detected on the surface of the solidified bond connections, which could be caused by the sudden cooling of the bond material.

The removal of the CBN grains starts with average laser power values of about 25 W corresponding to  $62.5 \mu\text{J}$  single pulse energy. Therefore, this value represents the threshold laser power value for the treatment of CBN grains. A laser trace generated with 25 W laser power and 10 mm/s scanning speed on vitrified bond CBN wheel is presented in Figure 9. Using higher laser powers, the cutting grains and the bond material will be effectively ablated. Figure 10 presents a trace generated with 50 W average laser

power and 10 mm/s scanning speed on vitrified bond CBN wheel, where a continuous groove is generated on the grinding wheel surface.

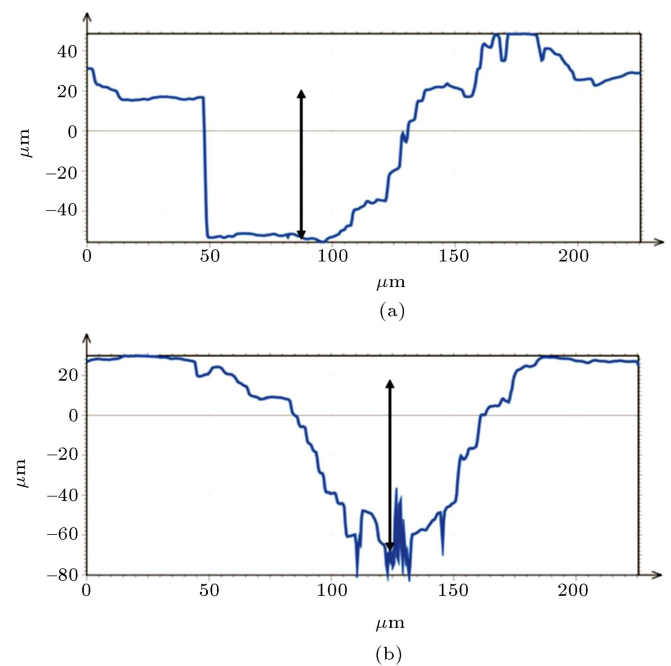
In order to survey the geometry of the grooves generated by laser, confocal microscopy of the surfaces was performed using a nanofocus surf microscope. Figure 11(a) and (b) present confocal images of two grooves generated on the vitrified bond CBN wheel with 10 mm/s scanning speed, and 20 W and 50 W average laser powers, respectively. The corresponding cross section profiles (“A-A” and “B-B”) are shown in Figure 12(a) and (b).

The approximate depths are shown with vertical

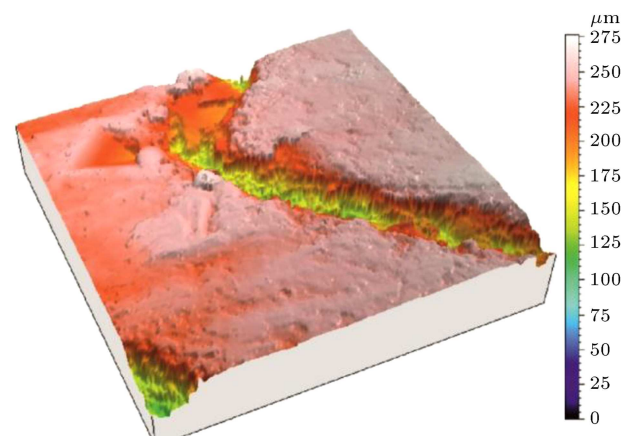


**Figure 11.** Confocal image of laser treatment on vitrified bond CBN wheel with 10 mm/s scanning speed: (a) 20 W average laser power, and (b) 50 W average laser power.

arrows in Figure 12. As the average laser power for generating the groove in Figure 11 is lower than the threshold power value for the ablation of CBN grains and higher than the threshold value for the ablation of the bond material, the CBN grain shown in the figure has not been treated while the bond material is scratched. The depths of the grooves presented in Figure 11(a) and (b) are approximately  $80\ \mu\text{m}$  and  $90\ \mu\text{m}$ , respectively. By increasing the average laser power up to more than two times (from 20 W in Figure 11(a) to 50 W in Figure 11(b)), the depth of the grooves in the bond material increases by only a few percent. In other words, the depth of laser treatment is nearly independent of the laser power and the function  $\tau$  in Eq. (6) is not a strong function of pulse energy. In order to generate grooves of the same depth in the resin bond grinding wheel, much higher scanning speeds would be needed, which is in



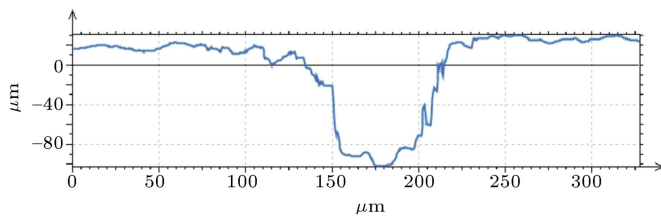
**Figure 12.** Cross section profiles of the laser grooves in Figure 11: (a) Cross: section “A-A”, and (b) cross section “B-B”.



**Figure 13.** Confocal image of laser treatment on resin bond CBN wheel with 200 mm/s scanning speed and 50 W average laser power.

agreement with the statement that the depth of grooves generated with a pulsed laser depends mostly on the scanning speed. Figure 13 presents the confocal image of a groove generated with 50 W average laser power and 200 mm/s scanning speed on the resin bond CBN wheel surface. The corresponding cross section profile is shown in Figure 14, where the depth of groove can be about  $100\ \mu\text{m}$ .

Figure 15 illustrates a complete resin bond grinding wheel, which was structured with 50 W average laser power and 200 mm/s scanning speed. The grinding wheel will be utilized and assessed in grinding process in the further steps of the research.



**Figure 14.** Cross section profile of the laser groove in Figure 13.



**Figure 15.** Resin bond CBN wheel structured with 50 W average laser power and 200 mm/s scanning speed.

**Table 3.** The threshold laser power values for the ablation of grinding wheel components.

Average laser power	Thermal model	Experiment
CBN grains	19 W	25
Vitrified bond	12 W	15
Resin bond	<1 W	2

The conducted analytical approach provides a correlation between the laser radiation parameters and the characteristics of the treated grinding wheel surface. The extent of cutting grain and bond material treatment is controlled by suitable values of laser power and scanning speed. The experimental results are in accordance with the analytical model. The threshold average laser power values for treatment of grinding wheel components are summarized in Table 3.

The laser power values predicted by the thermal modeling are slightly smaller than the experimental results. This discrepancy could be due to the energy loss of the laser beam through the optical components.

## 5. Conclusion

A picosecond laser device was integrated with a cylindrical grinding machine to perform in-place laser treatment of CBN grinding wheels. The present research can be summarized in the following points:

- It is suggested analytically and verified experimen-

tally that besides the thermal properties of the material, the maximum surface temperature depends directly on the laser pulse power;

- The critical average laser power values for the removal of the CBN grains, vitrified bond, and metal bond are found to be about 25 W, 15 W, and 2 W, respectively, which correspond to 62.5  $\mu\text{J}$ , 37.25  $\mu\text{J}$ , and 5  $\mu\text{J}$  pulse energy values, respectively. The energy gap between the threshold values corresponds to the laser power range in which bond material can be removed while the grains remain undamaged on the grinding wheel surface;
- The depth of thermal process penetration depends mostly on the laser scanning speed. The lower the scanning speed, the more time available for deeper temperature diffusion;
- It is possible to perform a wide range of conditioning processes (truing, dressing and structuring) of the resin and vitrified bond CBN grinding wheels with pulsed lasers when the parameters are chosen properly.

## Nomenclature

$c$	Specific heat capacity (J/kg $^{\circ}\text{C}$ )
$d$	Depth of laser treatment (thermal field penetration) (m)
$d_p$	Effective pulsed zone diameter (m)
$erfc$	Complementary error function
$f_r$	Laser pulse repetition rate (Hz)
$k$	Thermal conductivity (W/m $^{\circ}\text{C}$ )
$L$	Thermal diffusion length
$q''$	Heat generation per unit volume (W/m $^3$ )
$R$	Effective laser beam radius (m)
$T$	Temperature ( $^{\circ}\text{C}$ )
$T_i$	Initial temperature of material ( $^{\circ}\text{C}$ )
$T_{\max}$	Maximum surface temperature ( $^{\circ}\text{C}$ )
$t$	Time (s)
$t_0$	Laser pulse width (s)
$t_1$	Laser pulse relaxation time (s)
$\nu$	Laser scanning speed (m/s)
$W_{\max}$	Maximum single-pulse power (W)
$z$	Spatial coordinate in the depth direction (m)
$\alpha$	Thermal diffusivity (m $^2$ /s)
$\epsilon$	Thermal absorption
$\nabla$	Del operator
$\rho$	Density (kg/m $^3$ )
$\tau$	Groove depth empirical coefficient (-)



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