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# Tensile creep monitoring of basalt fiber-reinforced polymer plates via electrical potential change and artificial neural network

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### **KEYWORDS**

Tensile Creep Monitoring (TCM); Creep time; Electrical Capacitance Sensor (ECS); BFRP; FEM; ANNs. Abstract. In this research, Long-Term Tensile Creep (LTTC) failure in Basalt Fiber-Reinforced Polymer (BFRP) composites under ambient conditions was predicted and detected via an expert system in order to monitor the LTTC of BFRP laminated composites. This was accomplished by using a highly accurate, easy to use, and lowcost monitoring method incorporating an Electrical Potential Change (EPC) technique that employs an Electrical Capacitance Sensor (ECS) in conjunction with an Artificial Neural Network (ANN) to improve the process of detecting and predicting LTTC. A Finite Element (FE) simulation model for Tensile Creep (TC) detection was generated by ANSYS to obtain groups of data for the training of ANNs. The proposed method was then applied to minimize the extent of FE analysis in order to reduce the time required for the monitoring of creep behavior to a minimum. First, creep monitoring at different levels of TC ( $\%\sigma c$ ) as a percentage of Ultimate Tensile Strength (UTS) equal to 25%, 50%, and 75% was studied. Subsequently, the trained ANN was utilized to predict the creep behavior at different TC levels ( $\%\sigma c$ ) of 15%, 35%, 60%, and 85%, excluded from the FE data. The results showed excellent agreement between FE and predicted results.

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#### 1. Introduction

Basalt fiber is a relatively new and viable alternative material to the more commonly used composite fibers due to its good mechanical properties and lower production cost. For this purpose, as a key measure for assessing the performance and mechanical behavior of Basalt Fiber-Reinforced Polymers (BFRP), creep monitoring is important. It is particularly critical to detect long-term displacement levels and Creep Time (CT) for these new and advanced materials [1–6].

Due to the viscoelastic nature of the polymer matrix, the degradation of the modulus as a function of time, characterized by creep of polymer composites, has become a major concern for making structural parts using a polymer composite. Over the last three decades, considerable research has been conducted to understand the creep behavior of polymer matrices and

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composite structures. However, these studies on composite materials are very limited and inadequate [7].

According to literature, several techniques have been employed by various researchers to evaluate the mechanical properties of composite materials under creep tests. As reported in nearly all of the related published works, for tensile creep tests at maximum load, 50% of Ultimate Tensile Strength (UTS) has been considered. Researchers have also studied nonlinear viscoelastic models for composite materials at different levels of stress between 20 and 60 MPa in the temperature range of up to 90°C [8,9]. According to the findings of the above studies, it can be generally concluded that composite materials exhibit non-linear behavior at all applied levels of stress. In another relevant study on the creep behavior of a carbon/epoxy composite by using tensile and flexural creep testing, it was found that there was no creep rupture failure of the studied composites under a loading up to 77% of UTS for 1600 h at room temperature [10].

It is widely known that the performance and creep behavior of laminated composites are influenced by the properties of the matrix utilized in making the composites. The creep behavior of these composites not only maintains the desired form and protects the reinforcements against external attacks but also has an important role to play in the creep of composites. As discussed in the literature, the characterization of laminates has been studied based on the type and architecture of reinforcements [11] and is based on the epoxy matrix that influences its properties [12,13]. In this context, the creep behavior is strongly influenced by the viscoelastic properties of resin and the characteristics of fibers. In the case of Glass Fiber Reinforced Polymer (GFRP), the creep limit is 0.3UTS; in the case of Carbon Fiber Reinforced Polymer (CFRP) and Aramid Fiber Reinforced Polymer (AFRP), it is the rupture limit to the creep of 0.7UTS according to the recommendations of the American Concrete Institute (ACI) 440.4R04 [14].

Based on this literature review and to the best of the authors' knowledge, it is noted that no studies have investigated BFRP composites and the corresponding creep behavior. Therefore, this paper aims to provide more information on the effect of the creep behavior on the mechanical performance of BFRP laminated composites.

Fiber Reinforced Polymer (FRP) damage detection is complex in general and it is a difficult and expensive task to conduct during an inspection due to creep strain. The detection difficulty indicates the importance of developing an easy and economical technique for monitoring damage due to creep in FRP laminated composites [15–17]. The conventional Non-Destructive Test (NDT) methods, e.g., magnetic particle inspection or ultrasonic inspection techniques, are not able to detect creep damage because these methods are not able to detect damage prior to the formation of a creep crack in composite structures. Moreover, (a) traditional methods of sensing, which utilize strain gauges and vibration-based and piezoelectric type sensors, and (b) innovative monitoring techniques, which use advanced sensors such as optical fibers, generally incorporate sensors inside or outside of the structure and they are commonly expensive [18– 25].

In order to overcome these disadvantages, a number of other monitoring systems based on the intrinsically dielectric properties of materials have recently been studied. The methodology presented in this paper entails the measurement of Electrical Potential Change (EPC) of model parts using a row of electrodes mounted on the outer surface of structures without utilizing additional sensors. This row of electrodes measures the capacitance change due to the change in dielectric permittivity. Compared to other techniques such as traditional fiber optic sensors, the advantages of monitoring systems based on dielectric properties for damage detection in composite structures include the higher reliability and lower cost. Moreover, it is not possible to apply an intensive network of fiber optic sensors to large composite structures. Moreover, crack detection will fail if crack propagation does not intersect with the sensors. Furthermore, the installment of fiber optic sensors may result in initiating damage [26,27]. Conversely, Electrical Capacitance Sensor (ECS) offers more advantages such as low cost, fast response, low sensitivity to noise, higher safety, and continued operability under harsh environmental conditions [28-33].

Zhao et al. [34] studied a damage identification system for CFRP composite pipes using a 3D ECS model. They subsequently derived the transfer function of the system and applied it to an 'openloop' pipeline model using deep learning-based damage identification by extracting modal macro strains from dynamic excitations [35]. The proposed modulation method extended in this paper was tested in our previous work for crack monitoring due to fatigue load in laminated composite pipes [36] for identifying delamination [37,38] and detecting the mechanical properties and strength degradation of Glass Fiber Reinforced Epoxy (GFRE) composite pipes due to water absorption from internal hydrostatic pressure [39]. Altabey et al. [40] monitored the nano-delamination embedded in BFRP nano-pipes of EPC with Artificial Neural Network (ANN) for the first time and found that the proposed technique successfully assessed the nano-delamination embedded in composite nano-pipes within a low error band. In all of the previous works using ECS electrodes, the node potential values were measured before and after damage initiation.

To the best of our knowledge, no paper dealing with the electrical capacitance monitoring of FRPs under creep conditions can be found in the open scientific literature. Therefore, the aim of this work is to investigate the Long-Term Tensile Creep (LTTC) behavior at different levels of TC ( $\%\sigma c$ ) in BFRP laminated composite plates using ECS, whose strain and damage evolution under creep loading conditions is monitored by measuring the capacitance change due to the change in dielectric permittivity. In order to overcome the drawbacks of alternative monitoring systems, an ANN algorithm is utilized in this work to reduce the complexity of detecting and predicting the LTTC. First, the trained ANN is utilized to predict the Finite Element (FE) outcome of LTTC behavior and, then, it is subsequently used to predict the LTTC behavior at different levels of TC ( $\%\sigma c$ ) excluded from the FE data. The results show excellent agreement between FE results and those predicted by ANN.

# 2. Description of the sensor

ECS was used for the first time in the 1980s by a research group from the US Department of Energy to measure fluidized bed systems [41–43]. Subsequently, the technique has been further developed over the last 10 years. This approach has drawn the attention of the research community and has become significantly applicable to industrial process monitoring due to its low cost and reliable operation under harsh environments. ECS measures the capacitance change of multielectrode sensors due to the change of dielectric permittivity being imaged and, then, it reconstructs the cross-section images using the measured raw data with a suitable algorithm such as the algorithm used in this work via ANN. Electrical capacitance system includes a sensor and a capacitance measuring circuit, as shown in Figures 1 and 2. ECS consists of an insulating plate, measurement electrodes, and an earthed screen. The measurement electrodes are mounted on the top surface of the plate. The earthed screen is fitted between the electrodes to cut the electro line external to the sensor plate and reduce the inter-electrode capacitance. The earthed screen surrounds the measurement electrodes to shield external electromagnetic noise.

To increase the accuracy of ECS measurements, the factors affecting the sensitivity of ECS and its application domain have been studied. These studies have included the impact of structural material and inner dielectric permittivity [44–52] and the structural geometry (thinness ratio of structures) [53,54]. Altabey [55] studied the effect of environmental temperature on ECS working field for the first time and found that the ECS working field temperature caused substantial changes in the sensitivity of the ECS electrodes as well as its working domain.

#### 2.1. Geometric model

Figure 1 illustrates the LTTC failure monitoring system using the EPC with an ANN system. The model consists of seven electrodes that are fixed on one single side of the specimen surface, as shown in Figure 1. The plate used for the ECS in this research included a square plate with a length of 280 mm and a height of 15 mm. The plate was made of BFRP composite material with a row of seven electrodes that were mounted on a single surface of the plate, separated from each other by a 45 mm gap. Figure 3 illustrates the cross-section of seven electrodes of the ECS system configuration. The capacitance values of each pair of electrodes were measured via ECS, and the equivalent of node potential from the measured capacitance values was converted. The numbering order of electrodes system is shown in Figure 2. In ascending order, the electrodes were excited one by one so that if



Figure 1. Schematic representation of the monitoring method using Electrical Capacitance Sensor (ECS) and RS methods with an Artificial Neural Network (ANN).



Figure 2. Schematic representation of the measurement principle of an Electrical Capacitance Sensor (ECS).



Figure 3. Cross-section sketch of seven electrodes Electrical Capacitance Sensor (ECS).

one electrode was excited, other electrodes would be recorded at ground potential as detector electrodes (see Figure 2). The electrode capacitance,  $C_{ij}$ , can be calculated by using these charge measurements from Eq. (1):

$$C_{ij} = \frac{Q_{ij}}{\Delta V_{ij}},\tag{1}$$

where  $Q_{ij}$  is the charge induced on electrode j when electrode i is excited with a known potential.  $V_{ij}$  is the potential difference between electrodes i and  $j(\Delta V_{ij} = V_i - V_j)$ .

Thus, the number of independent capacitance measurements M = 21 using Eq. (2) is as follows:

$$M = \frac{N(N-1)}{2}.$$
 (2)

#### 2.2. FF simulation model

The structural properties of the BFRP laminated composite plate are shown in Table 1. Figure 4 represents the geometric model of the laminated composite plate structure. These BFRP composite properties were tested at the National and Local Joint Engineering



Figure 4. Element map of finite element mesh.

Research Center for FRP Production and Application Technology, Nanjing, China, a high-tech company specializing in the research and development, manufacturing, marketing, and technical assessment of highperformance fibers and composites.

Figure 4 shows that the laminated composite square plate has a length of 280 mm with 15 mm in height and the staking distribution to three plies is  $[0/90^{\circ}/0]_{s}$ . The thickness of each ply is 5 mm.

# 2.3. ECS Governing equations

The main target of derivations is to compute the capacitance matrix C from sensor parameters and structure material permittivity distribution  $\varepsilon(x, y)$ ; therefore, the first governing equation of ECS is Poisson's equation [34,36–39]:

$$\nabla \varepsilon(x, y) \nabla \varphi(x, y) = 0. \tag{3}$$

By solving Poisson equation (Eq. (3)) for the terminals Boundary Condition (BC) of ECS measurement system, the potential distribution inside the ECS  $\varphi(x, y)$ can be determined.

The other two sensor parameters include the electric field vector E(x, y) and the electric flux density D(x, y) calculated as follows [34,36–39]:

$$E(x,y) = -\nabla\varphi(x,y), \tag{4}$$

$$D = \varepsilon(x, y)E(x, y). \tag{5}$$

Gauss's law is used to find changes in the electrodes and the inter electrode capacitances. The law has been solved based on the following surface integral [34,36– 39]:

Table 1. Structural properties of the Basalt Fiber-Reinforced Polymer (BFRP).

Element	ρ	$\mathbf{E}\mathbf{X}$	EY	$\mathbf{E}\mathbf{Z}$	PRXY	PRYZ	PRXZ	GXY	$\mathbf{GYZ}$	GXZ
$\mathbf{type}$	$(\mathrm{kg}/\mathrm{m}^3)$	$(\mathbf{GPa})$	(GPa)	(GPa)				$(\mathbf{GPa})$	$(\mathbf{GPa})$	(GPa)
PLANE121	2700	96.74	22.55	22.55	0.3	0.6	0.3	10.64	8.73	10.64

Remark: BFRP is Basalt Fiber-Reinforced Polymers,  $\rho$  is material density, EX, EY, EZ are elastic moduli in the X, Y, and Z directions, respectively, GXY, GYZ, GXZ are shear moduli in the XY, YZ, and XZ planes, respectively, PRXY, PRYZ, PRXZ are Poisson's coefficients in the XY, YZ, and XZ planes, respectively.

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Tuble 2. Electrical properties.							
$\mathbf{Element}$	D	PERX	PERY	PERZ	RSVX	RSVY	$\mathbf{RSVZ}$
$_{\mathrm{type}}$	Permittivity	$(\mathbf{F}.\mathbf{m}^{-1})$	$(\mathbf{F}.\mathbf{m^{-1}})$	$(\mathbf{F}.\mathbf{m}^{-1})$	$(\mathbf{\Omega}.\mathbf{m})$	$(\mathbf{\Omega}.\mathbf{m})$	$(\mathbf{\Omega}.\mathbf{m})$
	Water	78.36	78.36	78.36	$1\mathrm{E}4$	$1\mathrm{E}4$	$1 \mathrm{E4}$
	Oil	3.0	3.0	3.0	3E11	$3  \mathrm{E11}$	3E11
SOL 1D123	$C_2H_5OH$	24.5	24.5	24.5	$7.4  \mathrm{E6}$	$7.4 \mathrm{E6}$	$7.4 \mathrm{E6}$
SOLID125	BFRP	2.2	1.32	1.32	0.01	0.01	0.01
	Electrode	$1 \mathrm{E} 10$	1E10	1E10	$1.75 \operatorname{E-6}$	1.75E-6	1.75 E-6
	Air	1.0	1.0	1.0	3E13	$3  \mathrm{E13}$	3E13

Table 2. Electrical properties.

Remark: Other parameters of the electrical property can be found in [60-62].

$$Q_{ij} = \oint_{S_j} (\varepsilon(x, y) \nabla \varphi(x, y) . \hat{n}) dS,$$
(6)

where  $(\nabla)$  is the divergence,  $(\nabla)$  is the gradient of parameters,  $S_j$  is a surface enclosing electrode  $j, \hat{n}$  is the unit vector normal to  $S_j$  and infinitesimal area dS on electrode.

#### 2.4. The FE model description

To investigate the effect of TC on the dielectric properties of the BFRP laminated panel, the FE analysis of the electric field intensity of laminated panel was designed using ANSYS ver.15 [56-59]. Suitable FEs were selected and employed to simulate BFRP properties, i.e., PLANE121 element is used to simulate structural property and SOLID123 is used to simulate electrical property. The elements were of rectangular 4-node type. Nodes and elements at each ply are 441 and 400 in number, respectively. The mesh size is 14 mm, the panel has fixed structural BCs on the one side, the tensile load is applied to the opposite side, and the other two sides are free. In electrical BCs, elements at each node have one degree of freedom (voltage). Tables 1 and 2 list all of the parameters required for multi-physics coupled field analysis. Other parameters of the electrical property can be found in [60-62]. The outcome of ANSYS software is the potential and electric field values (see Eq. (4)) at the element nodes. For each electrode, the potential BC is  $V_0 = 15$  V (RMS) applied one by one, such that one electrode is excited with 15 V (RMS) and other electrodes are kept at ground (V = 0) potential. To represent the natural propagation of electric field, the default BC of continuity  $(\hat{n} \cdot (D_1 - D_2) = 0)$  was maintained for the internal boundaries.

# 3. Description of the ANN algorithm

ANNs were used and developed first in the 1940s as information processing for non-linear and complex systems. Subsequently, the approach has been developed quickly over the last two decades and the approach has found a wide range of practical applications in numerous fields of engineering. One of the targets of ANN is to find complex and non-linear relationships between the input and output that cannot be found directly from experimental or numerical data. This is carried out by the proper training of ANN. Several types of ANN architectures have been used in different applications with different algorithms [63]. In this study, the ANN was applied to predict the electric potential differences between ECS electrode pairs.

# 3.1. ANN configuration

Due to several ANN configurations and the effects of these configurations on predictive accuracy and quality, it is necessary to describe the ANN configuration using a simple and easy code; this study used a general ANN configuration algorithm as follows:

$$\{N_{in}[N_{h1}N_{h2}]_e N_{out}\} \equiv \{7[39 \ 2]_2 1\}, \qquad (7)$$

where  $N_{in}$  and  $N_{out}$  are the element numbers of input and output parameters equal to 7 and 1, respectively, and e is the number of hidden layers equal to 2.  $N_{h1}$ and  $N_{h2}$  are numbers of neurons in each hidden layer equal to 39 and 2, respectively.

Note that the effectiveness of ANN depends on the related interconnections that increase based on the number of neurons in each hidden layer. These interconnections also represent the relationship between input and output. Thus, more training datasets are required to learn these relationships.

This study also needs to optimize the number of neurons in each hidden layer in ANN architecture [64].

#### 3.2. Feed-Forward Neural Networks (FFNN)

FFNN is the most commonly used ANs architecture type. It consists of a layer of input, a layer of output, and one or more hidden layers of neurons for full interconnection between input and output layers [65]. The hidden and output layer neurons use nonlinear activation functions such as linear transfer function (purelin (n)), Tan-Sigmoid transfer function (tansig(n)), or Radial Basis (Gaussian) transfer functions (radbas (n)). However, no activation function is used in the input layer since no computation is involved in that layer, and the data flows between layers in a feed-forward manner.

The training of FFNN continues until the Mean-Square-Error (MSE) between the input data and the network outcomes has reached a suitable value or after the completion of a pre-specified number of learning epochs. The MSE can be computed through the following equation:

$$MSE = \sum \left( (E_{i-j})_{nn} - E_{i-j} \right)^2 / 2, \tag{8}$$

where  $(E_{i-j})_{nn}$  is the predicted electric potential differences,  $E_{i-j}$  is the electric potential differences measured using the FE method, and n is the number of the FE measured data values.

# 4. Results and discussion

#### 4.1. Experimental validation

The experimental dataset used in this work to validate the proposed technique was adapted from Goertzen and Kessler [66]. The tests were conducted on composite laminated panels that were prepared with 2 plies of bidirectional woven Carbon Fiber-Reinforcement Epoxy (CFRE). Composite panels had the dimensions of 12 in×10 in (304.8 mm × 254 mm). The composite fiber volume fraction ( $V_f$ ) was measured to be between 35% and 40%. The thickness of the creep rupture specimens ranged from 1.1 to 1.2 in (27.94 to 30.48 mm). TC testing was performed using an in-situ creep rupture fixture (see Figure 5) developed by Lombart [67] at the University of Tulsa. Goertzen and Kessler [66] conducted the Tensile Creep Compliance (TCC)  $S(t) = 1/E(t) = \varepsilon(t)/\sigma(t)$  for the test specimens at 417 MPa (60.5 ksi) (65% UTS) and 496 MPa (72.0 ksi) (77% UTS).

To conduct a convergence investigation of the proposed technique, seven electrodes were fixed on the single side of the laminated panel surface with the same geometrical specifications. For the electrode model, the thickness of electrodes was 10 mm, the space between electrodes was 45 mm, and the BC of ECS was  $(V = V_0)$  with +15 V  $(V_0)$ . The EPC between electrode pairs was measured in various cases of TC  $(\%\sigma_c)$  as the percentage of UTS and was compared with the experimental results of Goertzen and Kessler [66].

Figure 6 shows the comparison between finite element data of the TCC (S(t)) and experimental data by Goertzen and Kessler [66] for the same laminated panel geometrical specifications, initial conditions, and CT.

From Figure 6, one can see excellent convergence between the FE and experimental data of the TCC (S(t)) with an average error of 2.2%. A small variation between the experimental and FE data exists because the finite element simulations ignore the fringe-field effects at the outer edges of the electrodes.

### 4.2. EPC technique for TCM

Seven electrodes were mounted on a single side of the specimen surface. For the electrode model, the



Figure 5. In-situ creep rupture fixture [67].



Figure 6. Comparison between Finite Element (FE) data of the Tensil Creep Comliance (TCC) (S(t)) and experimental data by Goertzen and Kessler [66].



**Figure 7.** Tensil Creep Comliance (TCC) (S(t)) of Basalt Fiber-Reinforced Polymer (BFRP) vs. Creep Time (CT), Tensile Creep (TC)  $(\%\sigma_c) = 25\%$ , 50%, and 75% Ultimate Tensile Strength (UTS).

thickness of electrodes was 10 mm, the space between electrodes was 45 mm, and the BC of EP was ( $V = V_0$ ) with +15 V ( $V_0$ ). The EPCs between electrode pairs were measured in various cases of TC ( $\%\sigma_c$ ) percentage of UTS equal to (25%, 50%, and 75%). From the measured data, TCC (S(t)) for each TC ( $\%\sigma_c$ ) percentage was obtained using the scripting capabilities in ANSYS.

Figure 7 shows the TCC,  $S(t) = 1/E(t) = \varepsilon(t)/\sigma(t)$ , for the BFRP Plate conducted at 135 MPa (25% UTS), 270 MPa (50% UTS), and 405 MPa (75% UTS).

According to Figure 7, it can be found that the compliance values of 25% UTS simulation were larger than those of 50% and 75% UTS simulation. This difference between compliance values resulted from fluctuations in the modulus of the composite material over the CT.



**Figure 8.** Effect of level of Tensile Creep (TC)  $(\%\sigma_c)$  on capacitance values of electrodes (pF).



Figure 9. Effect of Creep Time (CT) on capacitance values of electrodes (pF).

The corresponding reductions in modulus over the 1000-hour period ranged from a 20% reduction over 1000 hours at 25% UTS to a 68% reduction for 1000 hours at 75% UTS.

The exponential formula (9) to fit the FE results of TCC (S(t)) proved satisfactory by allocating applicable values to the Correlation Factor (CF) that were very close to unity. The values of four constants a, b,k, and h at  $\theta = 25^{\circ}$ C are shown in Table 3.

$$S(t) = ae^{bt} + ke^{ht}. (9)$$

Figures 8 and 9 show the 21 capacitance measurements  $(C_{ij})$  from 3D ANSYS model at different levels of TC  $(\%\sigma_c)$  and CT, respectively.

As shown in Figures 8 and 9, it can be observed that the effects of TC and CT on the capacitance

Tensile creep $(\%\sigma_c)$	CF	$a \times 10^{-11}$	$h  \times  10^{-7}$	$h \times 10^{-13}$	h	
level	C.F		0 × 10	κ χ 10		
**15%	0.9945	3.281	-4.365	-3.965	-0.008062	
$^*25\%$	0.9974	3.079	-4.082	-4.236	-0.007138	
$^{**}35\%$	0.9947	2.554	-4.064	-4.938	-0.007641	
*50%	0.9958	2.282	-3.906	-5.095	-0.008072	
**60%	0.9936	2.065	-4.429	-5.465	-0.007822	
*75%	0.9963	1.891	-4.733	-6.514	-0.007443	
**85%	0.934	1.645	-4.779	-6.967	-0.007336	
Avg.			-4.3368		-0.007644	
S.D			0.33831		0.000360	

**Table 3.** Creep compliance constants a, b, k, and h for Tensil Creep (TC) ( $\% \sigma_c$ ).

Remark: \*FEM data, and \*\*FFNN expected data.

measurement  $(C_{ij})$  distributions are introduced and a reduction in capacitance values between electrodes occurs. This reduction depends on the level of TC  $(\%\sigma_c)$  and CT. Following an increase in the level of TC  $(\%\sigma_c)$  and CT, the capacitance values are reduced.

# 4.3. FFNN design for ECS to study the LTTC behavior

FFNN is shown in Figure 10. The Neural Network (NN) configuration in this case is  $\{7[39 \ 2], 1\}$ , the first layer has tan-sigmoid neurons, and the activation function of the second layer is purely linear. The FFNN is trained in order to predict TCC (S(t)) by measuring values of  $\Im \sigma_c$ , CT,  $\theta$  in the input layer. First, the FFNN structure was applied to training the data of EPC at normal temperature ( $\theta = 25^{\circ}$ C). Figure 11 shows the difference between the FE data and the FFNN predicted data (TCC) at  $\theta = 25^{\circ}$ C at levels of TC  $\%\sigma_c$  equal to 25% and 75%. According to Figure 11 containing the NN results, the high accuracy of prediction in this case study can be observed. Figure 12 represents the difference between the FE data and the FFNN expected data at ( $\theta = 25^{\circ}$ C) at levels of TC  $\%\sigma_c$  equal to 50%. It is noted that FFNN results are in agreement with the FE data.

Table 4 shows the values of MSE (see Eq. (8)) between the FE and FFNN predicted and expected results at ( $\theta = 25^{\circ}$ C) in order to determine the best performance of the present network.

Neurall network Layer Layer Input U Output

Figure 10. Schematic illustration of Feed-Formal Neural Network (FFNN) design for the present study with input data Tensil Creep (TC) ( $\%\sigma_c$ ), Creep Time (CT),  $\theta$ .

# 4.4. Utilizing FFNN for predicting non-FE data

The main objective of ANN design is to predict non-FE data. In this section, the suggested FFNN is used to predict some non-FE data that have been



Figure 11. Comparison between the Finite Element (FE) data and Feed-Formal Neural Network (FFNN) predicted data at  $\theta = 25^{\circ}$ C.



Figure 12. Comparison between the Finit Element (FE) data and Feed-Formal Neural Network (FFNN) Expected data for Tensil Creep (TC) ( $\%\sigma_c$ )=50% at  $\theta = 25^{\circ}$ C.

Table 4. Mean Square Error (MSE) values.

$ heta=25^{\circ}\mathrm{C}$						
Data	Level of tensile creep, $\%\sigma_c$	MSE				
Prodicted	25%	8.4916 e-27				
1 lealitiea	75%	9.7817 e-26				
Expected	50%	6.5020 e-27				

excluded from the previous FE evaluation. FFNN is selected to be used for four cases where the level of TC ( $\%\sigma_c$ ) equals (15%, 35%, 60%, and 85%) for all the potential differences  $E_{i-j}$ . The previous three parameters, i.e.,  $\%\sigma_c$ , CT,  $\theta$ , are the input vectors for ANN, while the output is the signal vector for TCC (S(t)). Figure 13 shows the FFNN predicted results of the non-FE TCC (S(t)) in the BFRP laminated composite plate. The MSE of non-FE result is 2.0210E-26, 7.4194E-26, 3.1755E-26, and 1.7749E-25 for levels of TC  $(\%\sigma_c)$  equal to 15%, 35%, 60%, and 85%, All of the predicted FFNN data are respectively. plotted on the diagonal line in Figure 13. Based on these results, FFNN provides good predictions about non-FE data, even about the extrapolation of LTTC behavior in the BFRP laminated composite plate.

The application of the exponential formula  $S(t) = ae^{bt} + ke^{ht}$  has proved its viability by achieving acceptable values of the CF that are very close to unity. The results shown in Table 3 indicate that the proposed ANN is applied to the prediction of non-FE data.

# 4.5. The effect of TC ( $\%\sigma_c$ ) level percentage The analysis of the values of four constants, i.e., a, b, k, and h, including the variation of TC level ( $\%\sigma_c$ ), TCC (S(t)), and CT, as documented in Table 3, led to the following results:

1. At all TC levels  $(\%\sigma_c)$ , the tendency curve of TCC



Figure 13. Expected data of Tensil Creep Compliance (TCC) (S(t)) at four levels of Tensil Creep (TC)  $(\%\sigma_c)$  including 15, 35%, 60%, and 85% at  $\theta = 25^{\circ}$ C.

(S(t)) against CT at the TC level 15% has the highest TCC. The tendency curve at the TC level 85% has the lowest TCC, while the values of the other curves at the remaining TC levels range between the above two percentages with descending orders from 25% to 75%. Thus, it can be concluded that the difference between compliance values results from fluctuations in the modulus of the composite material over the CT. The corresponding reductions in modulus over 1000 hours period ranged from a 16% reduction over 1000 hours at 15% UTS to a 78% reduction for 1000 hours at 85% UTS;

- 2. The values of the constants (a and k) were found to be dependent on the TC level ( $\%\sigma_c$ ) for TCC (S(t)). As the TC level increases, the absolute value of (a and k) decreases, i.e., the TCC increases;
- 3. The values of the constants (b and h) at all TC levels ( $\%\sigma_c$ ) are negligible and may be considered constant. The average value (Avg.) of constants (b and h) was calculated and considered to be used at any TC level ( $\%\sigma_c$ ), as the corresponding Standard Deviation (SD) was found to have acceptable values, as shown in Table 3.

#### 5. Conclusions

In the present study, an Electrical Potential Change (EPC) technique was adopted as an expert system for monitoring the Long-Term Tensile Creep (LTTC) behavior at different levels of TC ( $\%\sigma_c$ ) to improve detection efforts in order to study the LTTC behavior. The application of the proposed technique lowers the cost and reduces the time duration of the Finit Element

(FE) analysis to a minimum with higher accuracy. The results obtained are given below:

- 1. Electric potential difference due to LTTC at different Tensile Creep (TC) levels ( $\%\sigma_c$ ) could be measured with multiple electrodes mounted on the outer surface of a Basalt Fiber-Reinforced Polymer (BFRP) plate;
- 2. The FE results showed good convergence with Feed-Forward Neural Network (FFNN) output results. This verified the accuracy and reliability of the proposed method, as shown in Table 4 and Figures 11 and 12;
- 3. Artificial Neural Networks (ANNs) could be employed as a method for simulating the Electrical Capacitance Sensor (ECS) from non-FE data of Tensil Creep Compliance (TCC) (S(t)) at TC levels ( $\%\sigma_c$ ) equal to 15%, 35%, 60%, and 85%;
- 4. At all TC (%σ<sub>c</sub>) levels, the tendency curve of TCC (S(t)) against CT at the TC level (15%) had the highest TCC. The tendency curve at the TC level (85%) had the lowest TCC, while the other curves at the remaining TC levels fell in between 15%–85%, with descending order from 25% to 75%. It can be concluded that the difference between compliance values results from fluctuations in the modulus of the composite material over the CT. The corresponding reductions in modulus over a 1000-hour period ranged from a 16% reduction over 1000 hours at 15% UTS to a 78% reduction for 1000 hours at 85% UTS;
- 5. The exponential formula  $S(t) = ae^{bt} + ke^{ht}$  proved its viability for the present study. It was found that the deviation of the constants (b and h) at different TC levels ( $\%\sigma_c$ ) and TCC (S(t)) was negligible, which might be considered to be constant;
- The values of the constants (a and k) were found to be dependent on the TC level (%σ<sub>c</sub>) and the TCC (S(t)) with high Correlation Factor (CF). As the TC level increased, the absolute value of (a and k) decreased, i.e., the TCC increased;
- 7. Finally, it can be concluded that the proposed approach provides a better understanding of the LTTC behavior at different TC levels ( $\%\sigma_c$ ).

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